From Nanocellulose to Wood Particles: A Review of Particle Size *vs.* the Properties of Plastic Composites Reinforced with Cellulose-based Entities

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This review article considers published evidence regarding effects of particle size on mechanical properties of plastic matrix materials filled with cellulose-based reinforcements. Cellulosic or wood-based reinforcements in plastic matrices can contribute to higher modulus, lower density, and less tendency to sag in comparison with the matrix phase by itself, while still allowing the resulting material to be cut or milled. Although cellulosic materials are generally too hydrophilic to adhere well to common thermoplastic materials such as polyethylene, such deficiencies can be overcome by use of compatibilizers, e.g. polyethylene-maleic anhydride. Recently many researchers have evaluated nanocellulose in plastic composites. The higher surface areas of nanocellulose generally imply a higher cost of compatibilizer to achieve good interfacial adhesion. This review first examines results of a large number of studies all involving highdensity polyethylene as the matrix. Then, to get a more detailed mechanistic view, studies are considered that compare different particle sizes of cellulose-based reinforcements within the same conditions of preparation of composites prepared with various matrix polymers. To summarize the findings, there does not appear to be any consistent and dependable advantage of using nano-sized cellulosic reinforcements when trying to achieve higher values of composite strength or modulus.

Keywords: Wood-plastic composites; Cellulose-based reinforcements; Size of filler particles; Nanocomposites; Tensile strength and modulus

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INTRODUCTION

There is strong interest, both in academic research and within industry, relating to the incorporation of cellulose-based reinforcements during the preparation of plastic composites to modify or enhance their performance. Such composites have the potential to be more weather-tolerant than wood in exterior applications (Wolcott *et al.* 1999). They also have the potential to be stronger than the base plastic in certain respects, such as the flexural modulus that determines the resistance to bending (Aji *et al.* 2011; Aggarwal *et al.* 2013). At the same time, the usage of under-utilized cellulose-based resources in composites has favorable environmental implications (Wolcott *et al.* 1999), including the sequestration of carbon (Bolin and Smith 2011) and the relatively low environmental impact of wood-based and cellulose-fiber-based products compared to some other commodities (Xu *et al.* 2008; Sathre and O'Connor 2010).

Developments in the field of cellulose-reinforced plastic composites have been reviewed (Thakur and Thakur 2014; Borah *et al.* 2016; Mohit and Selvan 2018). Biobased composites, in which the matrix as well as the reinforcements are derived from plant materials, including biothermoset resins, have received increasing attention (Reddy *et al.* 2010; Crosky *et al.* 2014; Chan *et al.* 2018; Pizzi 2018). Cellulose-based fibers (Lee *et al.* 2014; Ramamoorthy *et al.* 2015) can be obtained not only from wood, but also from plant sources such as cotton and flax. They also can be obtained from bacterial exudates, as well as animal sources in the case of tunicate nanocrystals. The aforementioned materials often can provide more eco-friendly alternatives by displacing the usage of petroleum-based, non-biodegradable plastics.

In the last two decades there has been intense interest, especially among academic scientists, in applications of nanomaterials (Berube 2006). It has been claimed, for instance, that nano-sized reinforcements, including nanocellulose, have the potential to achieve greater increases in composite strength in comparison to ordinary-sized reinforcements (Fu *et al.* 2008; Tanpichai and Wootthikanokkhan 2018). Also, nanocellulose appears to be able to increase the stiffness and other strength properties of plastic composites when present at relatively low levels (Puglia *et al.* 2015; Fortunati *et al.* 2016; Saikia *et al.* 2019).

On the other hand, nanomaterials are almost always more expensive than the larger materials from which they are produced. In the case of nanocellulose, two of the largest components of cost are often energy and chemicals (Delgado-Aguilar *et al.* 2015), and the balance between these two categories will depend on the type of nanocellulose being considered. When cellulosic materials are intended for use in plastic composites, one of the key costs categories to bear in mind is that of a compatibilizing treatment. Compatibilizers can involve either surface treatment of a cellulosic reinforcing material before its use or addition of a coupling agent to the matrix polymer before extrusion and compounding. As will become strikingly evident from literature to be considered in this review, compatibilizing treatments can play an essential role, especially in the case of nonpolar matrix materials such as polyethylene (George *et al.* 2001; Belgacem and Gandini 2005; Kumar *et al.* 2011). Accordingly, the very high surface area of nanomaterials necessarily inflates the associated costs.

The idea for the present review article first started as a matter of curiosity: Given that it often takes more energy and effort to prepare composites with smaller cellulosebased elements, the question can be asked as to what one can expect to get in return. An initial doubt about whether there is a benefit of using nanocellulose for plastic composites was expressed in an earlier review article (Hubbe 2017), but that publication presented little supporting evidence. A few examples from the literature pertaining to the topic were cited in a more recent conference presentation, titled "Think big: The case against relying just on nanocellulose to reinforce plastic composites" (Hubbe 2018). The present review article follows up with a more extensive examination of the published literature, with particular emphasis on effects of particle size in cellulose-based reinforcement of high-density polyethylene (HDPE) composites. This work also builds upon the important synthesis of information achieved in earlier review articles. Selected review articles, and their themes, are given in Table 1.

Table 1.	Review Articles	Dealing with	Cellulose and	Nanocellulose-base	b
Plastic Co	omposites				

Main Themes Discussed	References
Technology review of wood-polymer composites	Wolcott et al. 1999
Interfaces and natural fiber-plastic composites	George et al. 2001
Cellulose surface modification for adhesion to matrix	Belcacem & Gandini 2005
Cellulose-based nanocomposites	Hubbe et al. 2008
Cellulose fibril- and whisker-polymer nanocomposites	Gindl 2009
Cellulose nanofibers and nanocomposites	Eichhorn et al. 2010
Bast fibers and their composites	Summerscales et al. 2010
Wood flour-plastic composites	Kumar <i>et al.</i> 2011
Cellulose nanomaterials and nanocomposites	Moon <i>et al.</i> 2011
Use of recycled plastics in wood-plastic composites	Najafi 2013
Plastic reinforced with cellulose nanocrystals	Peresin et al. 2013
Numerical simulation of wood-plastic composites extrusion	Yuan & Chen 2013
Extruded and solvent-casted cellulosic composites	Forunati <i>et al.</i> 2016
Surface treatments of natural fibers for composites	Verma & Jain 2017
Fracture toughness and impact strength	Al-Maharma & Sendur 2019
Flax fiber reinforced plastic composites	Ramesh 2019

THEORETICAL BACKGROUND

Why Two Phases Can Be Better than One

The motivating principle of composites is to achieve properties that cannot be reached using either of the constituent materials by itself (Wolcott *et al.* 1999). Often the goal is to achieve higher strength attributes, taking advantage of an inherent higher modulus of elasticity of a reinforcing (filler) material. As noted by Kalaprasad *et al.* (1997), the strength properties of plastic composites reinforced by relatively short cellulosic fibers generally depend on the fiber loading, any orientation of the fibers, the lengths of the fibers, and the strength of adhesion between the fibers and the plastic matrix. The cited work discusses the main theories that have been developed to account for mechanical properties of composites. In general, the models are able to explain the main features of experimental findings. Another common objective, which lies outside of the scope of the present review, would be to lower the cost of the resulting product by filling the plastic with a less expensive material, while still meeting the requirements for the product (Lightsey *et al.* 1977; Toupe *et al.* 2014).

Mechanisms of Failure

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The strength of a composite is generally judged by how it deforms and how it ultimately fails when subjected to forces and stresses under defined geometries and conditions (Jenkins 1998; Campilho 2016). In particular, both the tensile modulus and the shear modulus can be evaluated by determining the ratio of stress to strain under conditions that are usually selected to be within a linear range, *i.e.* stress levels that do not cause damage. By contrast, the term "strength" is often applied to tests that evaluate the maximum tensile stress or flexural stress leading to breakage (Fu *et al.* 2008; Ku *et al.* 2011; Dhakal *et al.* 2015). These quantities, along with the strain associated with tensile breakage, will be a main focus of this review when considering experimental findings.

Breakage represents a dramatic form of failure, and tensile breakage is the simplest case to consider. In brief, tensile breakage can involve either failure of the reinforcing elements (*e.g.* cellulosic fibers or nanocellulose), breakage of the matrix material, debonding between the matrix and the reinforcement, or debonding within agglomerates of the reinforcing elements. Once a crack is initiated, the process often proceeds in an out-of-control manner, following paths of weakness in the structure. Some references to each of the mentioned events, related to cellulose-based plastic composites, are given in Table 2. None of the cited articles report specific evidence of failure initiated by fracture within a plastic matrix phase, except when the plastic by itself was being evaluated as a control.

Table 2.	Mechanistic	Evidence fron	n Various	Studies (of Cellulose	-based	Plastic
Compos	ites						

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Type of Mechanistic Evidence	Selected References
Fracture of the cellulose-based reinforcement	Summerscales et al. 2010; Alavi et al. 2013;
	Fonseca-Valero et al. 2015; He et al. 2016
Debonding between the matrix & reinforcement	Alavi et al. 2013; He et al. 2016; Mu et al.
	2018; Simao <i>et al.</i> 2019
Debonding within agglomerates of cellulose-	Ismail & Shafiq 2016; Lewandowska et al.
based reinforcements	2018; DiLoreto et al. 2019
Large increases in strength when using	George et al. 2001; Fu et al. 2008; Farsi
compatibilizing agents	2012; Alavi et al. 2013; Durmus et al. 2019

The quality of bonding between a cellulosic surface and a plastic matrix material tends to be of critical importance. Gaugler *et al.* (2019) recently showed that such bonds can be evaluated quickly and precisely by preparing sandwich-type composites. Specimens consisting of wood in combination with various thermopolymers were prepared over a range of temperatures, followed by *in situ* mechanical testing. The polarity of the plastic was found to be important. Fluorescence microscopy visualization of laminate composites bond lines revealed the ingress of thermopolymer into the pores of the adjacent wood phase, which was found to be temperature-dependent. The degree of ingress of polymer into the pores of the wood was indicative of effective adhesion between the phases (Luedtke *et al.* 2019; Grigsby *et al.* 2019).

Aspect ratio issues

The aspect ratio of a particle can be defined as the quotient of its length divided by its thickness. The relationship between aspect ratio and the strengths of composite materials has been discussed in review articles (Tucker and Liang 1999; Puglia *et al.* 2015; Fortunati *et al.* 2016). A more effective transfer of stress from the matrix to the fiber is

expected when the aspect ratio is higher (Frone *et al.* 2011). Stark and Rowlands (2003) attributed greater reinforcing effects of wood fibers in comparison to wood flour in polypropylene composites to the higher aspect ratio of the fibers. It has been found that the high aspect ratio in addition to the high surface area of nanofibers from woody material usually make a positive contribution to strength in the composites (Fortunati *et al.* 2016). Xu *et al.* (2013) used similar reasoning to account for a greater contribution to strength from nanofibrillated cellulose in comparison to cellulose nanocrystals. Similarly, Gozdecki *et al.* (2015) used aspect ratio to the strength of composites. Khonsari *et al.* (2015) attributed a greater contribution of ground shavings of wood flakes to flexural modulus to their relatively high aspect ratio in comparison to wood flour. Sapkota *et al.* (2017) found that the effects of aspect ratio on composite properties were strongly influenced by the contribution of those particles having the highest aspect ratio.

Though higher aspect ratio often decreases the probability that the reinforcement will detach from the matrix during breakage, one can also generally expect a greater extent of agglomeration in the course of preparing the composite (Hubbe et al. 2017). Furthermore, reinforcements that are long relative to their diameter or thickness will generally increase the viscosity of the mixture, making it more difficult to process. During high-shear processing, as with twin-screw extrusion, reinforcement particles having a high aspect ratio are often broken, yielding a mixture that has a much lower distribution of aspect ratios (Bouafif et al. 2010; Thumm and Dickson 2013; Soccalingame et al. 2015). Li et al. (2014) reported a case in which addition of a dispersion agent to HDPE succeeded in getting a better mixing of cotton nanofibers, while at the same time resulting in a greater degree of shortening during processing. Another important factor is the deviation from rod-like shape, even in cases where cellulose-based particles might initially appear straight. Newman et al. (2014) showed that cellulosic fibers tend to develop curves and kinks during injection molding, whereas glass fibers tend to remain rod-like. As a consequence, cellulose-based fibers, when used as fillers in injection molding of composites, would not be expected to match the theoretically predicted modulus values. Rather, the theoretical models tend to over-estimate the strength of cellulose fiber-reinforced composites (Newman et al. 2014). As will be seen in later sections of this article, composites prepared with nanofibrillated cellulose sometimes have been reported to achieve relatively high gains in mechanical properties, compared to similar levels of other reinforcing particles. The very high aspect ratio of nanofibrillated cellulose is expected to have been an important factor to explain such results.

Positive Aspects of Using Cellulose-based Nanoparticles

If indeed there are advantages to using smaller, even nano-sized reinforcing elements in a composite, what could be the theoretical basis used to explain or predict such benefits? One such theory involves effects of the reinforcements on the degree of crystallinity of the matrix material when it cools from a melt. Various studies have reported evidence for increased crystallinity in the presence of nano-sized cellulose-based reinforcements (Mokhena and Luyt 2014; Boran *et al.* 2016; Sun 2018; Pereira *et al.* 2019; Peric *et al.* 2019; Wang *et al.* 2019a). Such effects would be expected to be related to the surface area of the reinforcements, which is much higher in the case of nanomaterials. Inducing additional crystallinity within the plastic phase generally provides a higher elastic modulus, leading to greater stiffness (Krishnaiah *et al.* 2017).

Some potential advantages of using very small reinforcing particles fall outside of mechanical properties, which is the primary focus of this review. For example, the tiniest of cellulosic particles, the cellulose nanocrystals (Samir *et al.* 2005; Eichhorn *et al.* 2010; Moon *et al.* 2011), have a very high level of crystallinity (*e.g.* 54-88%, Moon *et al.* 2011), and this could be an advantage sometimes in terms of structural stability or insensitivity to moisture (Tyagi *et al.* 2019). In addition, some potential applications such as 3D printing may require thermoplastic formulations to pass through very small openings (Siqueria *et al.* 2017). Certain nanocomposites can achieve very high levels of transparency, which may be beneficial in some applications (Eichhorn *et al.* 2010; Soykeabkaew *et al.* 2012; Xu *et al.* 2013; Nair *et al.* 2019; Saikia *et al.* 2019). While dependent on the formulation, highly transparent composites can be prepared when using nanocellulose particles for reinforcement (Nair 2019).

Negative Aspects of Using Cellulose-based Nanoparticles

Some potential downsides of using nanomaterials for reinforcement of plastic composites were already detailed in the Introduction. For example, nanocellulose products – and even fine-ground particles or flours – tend to involve higher energy input, along with higher costs of preparation (de Assis *et al.* 2017). If there is a need to treat the surfaces or to add a compatibilizing agent to the matrix, then intuitively the required amounts of these agents will tend to be higher when the surface area of the reinforcing particles is increased, as in the case of nanoparticles (Hubbe 2017). In support of this concept, Tarrés *et al.* (2019) observed that specific amounts of coupling agent were needed in polypropylene systems reinforced with henequen strands of different size in order to achieve maximum strength. There are also some practical challenges inherent in working with particles that are too small to be collected easily by filtration. Thus, centrifugation is widely used to increase the solids content of nanocellulose suspensions (Wang *et al.* 2012). It also can be a challenge to monitor and control the properties of cellulose nanoparticles, especially in the case of highly fibrillated cellulose products, due to their tiny dimensions and complex structure (Lavoine *et al.* 2012; Kangas *et al.* 2014).

Positive Aspects of Using Macroscopic Cellulose-based Reinforcements

When considering wood as a material for construction, it is important to keep in view the full range of dimensions, starting from molecular, proceeding through nano-range structural arrangements of cellulose, hemicelluloses, and lignin within small sections of cell walls, then proceeding to the layered arrangements of fibrils within fibers, and then to the manner in which fibers are bonded together to constitute wood. These essential issues are addressed in textbooks of the field (Lewin and Goldstein 1991; Kettunen 2006). At each scale of observation, the structural components of wood are arranged in a well-optimized pattern that is capable of transferring stresses and strains in an efficient manner, thus achieving a remarkable combination of material strength, stiffness, and toughness. For example, the microfibril angles in the different sublayers of a fiber cell wall are arranged to achieve a favorable tensile modulus response, while also protecting against the collapse or bulging of an individual fiber (Fig. 1). All of these parts are already essentially "glued together" in their natural state. This being the case, it would seem logical to use the largest practical pieces of the original wood as the starting point for preparation of a composite material.



Fig. 1. Illustration of the layered structure of a typical fiber within wood. Note that "P" stands for "primary" and that "S" followed by a number indicates parts of the "secondary" layer. Lines drawn on the P, S₁, S₂, and S₃ layers indicate the predominant orientations (or lack thereof) of the microfibrils.

Relatively large pieces of cellulosic material, including wood particles and fibers, have a much lower outer surface area per unit mass in comparison to nanocellulose. As an extreme example, a cellulose nanocrystal has a specific surface area of *ca*. 500 m²/g, whereas dried kraft fibers have surface areas of the order of magnitude of only 1 m²/g (see sample calculations in the Appendix). The much lower specific surface area of relatively large fibers or particles makes it much more reasonable to consider strategies that involve surface modification. As shown in review articles, the hydroxyl groups at cellulosic surfaces provide opportunities for various chemical modifications, leading to better adhesion to a hydrophobic matrix (Belcacem and Gandini 2005; Hubbe *et al.* 2015; Verma and Jain 2017).

Table 3 lists some of the major fiber surface modification chemistries that have been demonstrated for improving adhesion within a plastic matrix. The first two categories considered in the table involve chemical derivatization of the cellulosic surfaces before preparation of the composite. By contrast, the third and fourth categories (identified with an asterisk) involve addition of a compatibilizing agent to the plastic material before compounding or hot-pressing. The two main compatibilizing agents that are used this way, especially with polyethylene and related matrix types, are maleic anhydride polyethylene (MAPE) and maleic anhydride polypropylene (MAPP). In these two cases, the covalent reactions with the cellulosic surfaces generally take place only after the reinforcing material has been added to the polymer, either as a melt or as material to be heated during an extrusion process. An exception to this rule was the work of Zhang *et al.* (2019a). In that study the surfaces of cellulosic nanocrystals (CNC) were individually modified with MAPP. Such treatment led to large improvements in the strength of composites formed with an acrylic copolymer.

Table 3.	Surface Modification Chemistries Shown to be Effective to Improve
Adhesion	of Cellulose-based Reinforcements within Various Plastic Matrix Types

Type of Chemical	References
Treatment	
Silane (trimethoxy-	Raj and Kokta 1991; Cui et al. 2008; Frone et al. 2011; Mokhene &
silane compounds)	Luyt 2014; Farsi 2012; Nishitani <i>et al.</i> 2016; Chen <i>et al.</i> 2017; Fang
	<i>et al.</i> 2017; Gao <i>et al.</i> 2017, 2019; Zhou <i>et al.</i> 2017; Rashno <i>et al.</i>
	2018; Fathi <i>et al.</i> 2019, Ikladious <i>et al.</i> 2019; Liu <i>et al.</i> 2019
Esterification (including	Zhang et al. 2009; Taib et al. 2010; Ben Mbarek et al. 2013; Cavdar
acetylation and	<i>et al.</i> 2014; Cetin <i>et al.</i> 2015; Hung <i>et al.</i> 2016; Biswas <i>et al.</i> 2019;
monomeric alkyl esters)	John <i>et al.</i> 2019
Maleic anhydride	Bouafif <i>et al.</i> 2010; Sewda & Maiti 2010; Taib <i>et al.</i> 2010;
polyethylene (MAPE)*	Zabihzadeh et al. 2010; Najafi & Khademi-Eslam 2011; Yao et al.
	2011; Li 2012; Petchwattana et al. 2012; Ben Mbarek et al. 2013;
	Gallagher & McDonald 2013; Li et al. 2013; Hemmasi 2013; Liu et
	al. 2013; Pollanen et al. 2013; Tisserat et al. 2013a,b; Schirp et al.
	2014; Tazi et al. 2014; Tong et al. 2014; Migneault et al. 2015;
	Roumeli et al. 2015; Tisserat et al. 2015a,b; Boran et al. 2016; Tufan
	& Ayrilmis 2016; Cisneros-Lopez <i>et al.</i> 2017; Hosseini <i>et al.</i> 2017;
	Zhou et al. 2017; Huang et al. 2018; Kajaks et al. 2018;
	Lewandowska et al. 2018; Mu et al. 2018; Zhang et al. 2018; Biswas
	<i>et al.</i> 2019; Ihamouchen <i>et al.</i> 2019; John <i>et al.</i> 2019
Maleic anhydride	Cui et al. 2008; Zhang et al. 2009; Kaymakci & Ayrilmis 2014; Toupe
polypropylene (MAPP)*	et al. 2014; Soccalingame et al. 2015; Chang et al. 2016; Elloumi et
	<i>al.</i> 2016; Gomes <i>et al.</i> 2018; Igarashi <i>et al.</i> 2018; Durmus <i>et al.</i>
	2019; Simao et al. 2019; Xu et al. 2019

*Note: The typical way that polyethylene- or polypropylene grafted maleic anhydride (MAPE or MAPP) is employed is by mixing it with molten thermopolymer. Then, during the compounding operation, the cellulose-based surfaces are able to form esters by reaction with the anhydride groups of MAPE or MAPP.

Concerns about Using Macroscopic Fibers

While the above discussion provides reasons to suggest that "bigger is better," with respect to cellulose-based reinforcements for typical plastics, there are some contrary arguments to this assertion. Perhaps the most important of these is the fact that a relatively large reinforcement particle can act as a relatively large defect and point of crack initiation within a composite (Alavi *et al.* 2013). Such effects are especially to be anticipated when there is poor interfacial adhesion, *e.g.* when using unmodified wood fibers as a filler in a similarly unmodified polyolefin matrix.

There are cases in which larger cellulose-based fibrous elements may be inherently weaker than their smaller pieces. This can happen, for instance, in the case of compound fibers that are comprised of smaller elements. As discussed by Zimniewska *et al.* (2011), some bast fibers are bundles of elementary fibers held together by pectin and other compounds; such junctions can serve as weak points when the bundles are placed under stress. Such a mechanism might explain the results obtained by Nishitani *et al.* (2016). These researchers found that the positive effect of initial length of hemp fibers, with respect to the tensile properties of polyamide composites, did not increase much after the initial length was increased beyond 1 mm. Presumably, beyond that point the junctions between elementary fibers may have become separated in response to the tensile forces applied to the composites. Similarly, Tarrés *et al.* (2019) reported a linear negative relationship between fiber length and intrinsic strength of henequen strands; again, this finding is consistent with the existence of weak points in compound fibers. Rong *et al.* (2001) go

into more detail about how separations occurring between primary bast fibers in a multifiber strand (sisal bast fibers) provided a mechanism of preferential breakage after their formation into sisal-epoxy composites.

As shown by Di Guiseppe *et al.* (2017), the compounding of fibrous reinforcements within a polymer melt can generate large stresses. The accumulation of such stresses acting at the surface of cellulosic particles can be enough to cause them to fracture. This mechanism can explain many reports in which the average length of cellulosic fibers became greatly reduced in the course of extrusion to form a composite (Bouafif *et al.* 2010; Fonseca-Valero *et al.* 2015; Soccalingame *et al.* 2015). In such situations, potential advantages that one might have been expecting from larger particles will have been lost. It has been reported that larger cellulosic particles tend to experience more severe damage during extrusion to form a composite (Jaya *et al.* 2016).

Some other potential disadvantages of larger reinforcing elements in composites concern the quality or appearance of the surface. For instance, it has been reported that larger cellulose-based reinforcements (wood particles) yielded rougher surfaces of composites in comparison to when individual fibers were used as reinforcements (Wechsler and Hiziroglu 2007). Likewise, formation of agglomerates of reinforcing elements within composites can manifest itself as a rougher surface of the composite (Tisserat *et al.* 2013a).

Expectations Based on Modeling

It is inherently difficult to model the stresses and strains associated with a threedimensional composite filled with diverse particles aligning in multiple orientations. However, some insights have been provided by models that involve simplifications. Tucker and Liang (1999) considered the most relevant models, based on a set of simplifying assumptions. These included linear elastic behavior of both the plastic matrix and the reinforcing particles, isotropic behavior of the matrix, axisymmetric shape of the reinforcing particles, and perfect bonding at interfaces, with no slippage. None of the models considered by these authors showed any expected dependency of elastic modulus of the composite on the size of the reinforcement. Thus, none of the models considered can be used to predict any advantages of using nano-sized reinforcements, assuming that the shapes of different-sized particles are similar.

A possible objection to relying on the kinds of models considered by Tucker and Liang (1999) is that those models do not consider potential effects at the nano-scale. An argument could be made, for instance, that the very high surface areas associated with nanocellulose could give rise to effects that would not be otherwise predicted. For example, there might be contributions from forces acting between particles within a medium. Thus, the previously mentioned expectation of a "nano-advantage" (Fu *et al.* 2008; Tanpichai and Wootthikanokkhan 2018) ultimately deserves to be evaluated on an experimental basis, as in the following section.

EXPERIMENTAL FINDINGS

Strategy: Consider Big Picture First, Then Details

Since there is a very large amount of scientific literature related to plastic composites reinforced by cellulose-based materials, it was necessary to make some strategic decisions on how to organize this review of the literature. It was decided to divide the discussion of experimental findings into two parts. The first part attempts to answer

some broad questions, based on selection of a common matrix polymer. The idea is that by focusing on a large number of studies all evaluating the same kind of polymer, it can be possible to look for characteristic differences in the nature of results that can be obtained with addition of very different sizes of cellulose-based reinforcements. The second step was to focus on specific publications where the researchers happened to study effects of particles having different sizes under their own fixed conditions of matrix type and processing conditions.

Why hybrid composite studies were not included in the data set

There is much current interest in the concept of hybrid composites, *i.e.* composites in which two or more kinds of reinforcement are used simultaneously within a matrix. However, such combinations were not included in the publications considered for the present analysis. The primary reason was to avoid additional complexity. As will be shown, it is already a difficult challenge to find clear evidence for dependencies of composite properties on the dimensions of reinforcing particles, and properties of hybrid composites are yet harder to account for. An earlier review of hybrid composites (Hubbe 2017) found only a few examples of studies giving clear support for the idea that a combination of two different types of reinforcement (including different sizes or shapes) can be beneficial for composite strength. Most results of work involving such hybrid composites either can be understood by "rule of mixtures" estimates or are merely inconclusive with respect to the theory of hybrid composites (Philips 1976).

Coupling agents and surface modification regarded as a second focus

Although coupling agents and surface modifications and treatments were not the main issues considered during the formulation of the present study, the review of the literature immediately made it apparent that, just like the "elephant in the room," issues related to surface adhesion can play major and even dominant roles in many situations of interest (George *et al.* 2001; Belcacem and Gandini 2005; Verma and Jain 2017). Accordingly, in many of the tables and figures provided below, table entries and data points will be presented with an indication of whether or not there was some form of compatibilization employed. It must be borne in mind, however, that the range of reported studies is extremely diverse. Hence, even if a given study indicates that "yes," there was a coupling agent or surface modification, one must be open to the possibility that the treatment might have been extremely ineffective or have had negative effects in a specific piece of published research.

Largest dimension used as the key measure

A third key choice that was made to simplify the analysis that follows was to consider the largest dimension of the reinforcing particle as the only measure of particle size. This was done because much of the reported data is based on use of sieving through standard mesh screens as a way to obtain particles within specified size ranges. Only in a minority of cases do such publications provide sufficient information to estimate aspect ratios and fiber widths. Fuller comprehensive analysis of such issues is left as an opportunity for future investigations.

Part A: Studies Dealing with a Selected Common Polymer Matrix

Selection of a common plastic matrix material for data mining

The analyses and comparisons discussed in this section are based on an arbitrary selection of a single kind of polymer matrix. By focusing on a large number of studies all dealing with only one class of plastic, the goal was to try to address some broad questions, such as the following:

- When comparing results over a very broad range of reinforcement sizes, is there an observable effect of particle size on the strength attributes of composites?
- In addition to any effects attributable to particle size, is there an observable influence of coupling agents or surface modification of the cellulose-based reinforcement particles?

High-density polypropylene (HDPE) was selected as the polymer matrix for this part of the study. The choice of HDPE as a focus area is justified not only by its very widespread use as an unmodified polymer material, but also because of the large number of studies that have been conducted involving cellulose-based reinforcements. HDPE also presents a specific challenge that is of great interest to researchers: The low surface energy of HDPE surfaces tend to bond poorly to unmodified cellulosic surfaces (Hou *et al.* 2017; Sakakibara *et al.* 2017; Zhang *et al.* 2018; Gao *et al.* 2019), so that a majority of related published research includes evaluating effects of compatibilizing treatments. Another characteristic of interest is HDPE's moderately flexible nature. Tensile modulus values of HDPE have been reported most often in the range 0.2 to 2 GPa (see Table A in the Appendix), which is much lower than the values reported for crystalline cellulose (CNC), which has been reported as 138 GPa (Sakurada *et al.* 1962). This contrast implies that the combination of HDPE and cellulose-based reinforcements at least has the potential to achieve elastic modulus values that are significantly higher than that of the matrix polymer by itself.

The goal of this analysis is to determine whether, above the variations associated with other factors such as particle shape and orientation, it is still possible to detect statistically significant trends in the data. Four approaches were employed in the Part A in an attempt to deal with inherent variations in the details of different studies. First, all the data considered here deals with the same nominal matrix material, HDPE. Second, since a very large number of studies have been focused on HDPE, the number of individual results to be compared likewise is about as high as can be found in the literature at this point. Third, to minimize the impact of differences in the matrix properties in the compared studies, the data were compared in the form of ratios of observed data relative to the unfilled corresponding matrix value reported in a given study. Fourth, the range of particle size considered in this work was very large.

Before considering experimental data, it is important to note that large differences can be expected in the matrix polymer, even when the designation "HDPE" was used in each of the systems considered. Factors such as molecular mass distributions, degrees of crystallinity, and purity can be expected to show differences. Processing conditions also can be widely different among nominally similar experimental studies. For example, the rate of cooling a composite after compounding, which can be different from study to study, can be expected to affect the crystallinity and related properties of the polymer (Supaphol and Spruiell 2002; Kundu *et al.* 2003). The methods of processing and formation of the composites were grouped together, rather than being isolated. As a way to minimize the

influence of differences in matrix properties, the data to be considered in this part are shown mainly on a relative basis, comparing the composite properties to those of the corresponding specimen of unfilled HDPE in each of the studies.

Tensile strength

Combined data obtained from approximately 50 separate published studies are presented in Fig. 2, all based on experiments in which cellulose-based elements were used to reinforce HDPE matrices.



Fig. 2. Published results for the relative change in tensile breaking strength, compared to that of the HDPE matrix employed in each study, *vs.* the length (usually determined by screening) of a cellulose-based reinforcement. Circle symbols correspond to untreated cellulosic particles. Squares correspond to use of a coupling agent or surface agent intended (by the given authors) to improve the adhesion.

The vertical axis represents the ratio of the observed tensile breaking stress to that of the neat, unfilled HDPE by itself in each respective study. The horizontal axis corresponds to the reported size or length (often based on the reported mesh size) of the particles. Note that the scale of the horizontal axis is logarithmic, showing a range of about five factors of ten. Experimental details are listed in Table A, which appears in the Appendix. The contents of reinforcements, by mass, were generally in the range of 20 to 60% for most reported results. However, lower amounts, usually in the range of 1 to 10%, were most commonly used for systems reinforced by nanocellulose (CNC and NFC).

Two main conclusions can be drawn from the plotted results in Fig. 2. First, there was no discernable trend for tensile breaking stress as a function of the logarithm of particle size. Rather, there was a very large range in relative tensile strengths observed within each size range within the plotted data. A linear regression of all the data (110 observations) yielded an R^2 value of 0.00012 and a slope of having a 95% confidence interval of -0.38 to +0.43. Thus, there was no statistically significant relationship.

The second main conclusion is that those systems employing a compatibilizer (*i.e.* either a coupling agent or specific treatment of the cellulosic surface with the goal of making it more compatible with the HDPE) generally yielded higher strength. On a quantitative basis, the average value of relative tensile strength for all of the data was 1.21, while it was 1.40 when considering just the compatibilized systems.

A further linear regression calculation was carried out considering just the 59 observations associated with compatibilized systems. In this case the R^2 value for the linear relationship between tensile strength and size was 0.00180 and the slope of the regression line had a 95% confidence interval between -0.58 and 0.42. Thus, for the compatibilized system, there still was no statistically significant relationship between tensile breaking strength and the logarithm of particle size in HDPE composites.

Tensile modulus

Results for the relative tensile modulus (normalized to the modulus of the neat HDPE in each study) *vs*. the logarithm of particle size are shown in Fig. 3.



Fig. 3. Published results for the relative change in tensile modulus, compared to that of the HDPE matrix employed in each study, *vs.* the length (usually determined by screening) of a cellulose-based reinforcement

As above, there was a large range of tensile modulus values at each size range. A linear regression of all the data (98 observations) gave an R^2 value of 0.047 and a slope having a 95% confidence interval of +0.0130 to +0.267. Given that this whole range is (sometimes just barely) positive in value, this result implies a significant increase in relative tensile modulus with increasing particle size. However, the weakness of this statistical fit must be emphasized. When considering only the compatibilized systems, the R^2 value was 0.0099 and the slope did not show a statistically significant trend with respect to particle size.

Regarding the effects of the compatibilizing treatments, untreated reinforcements yielded an average increase in tensile modulus of 2.17 relative to the neat HDPE, whereas

the systems with coupling agents or surface treatments yielded an average relative increase of 2.41.

Tensile elongation

Results for tensile elongation at breakage, relative to the elongation of the respective HDPE samples alone, are shown in Fig 4. Three things are worth noting about these results. Firstly, there were many data points very close to zero at the bottom of the plot, which is reflective of the inclusion of particles inducing very brittle systems. Secondly, there were a few systems, representing very different particle sizes, in which the elongation at break was about as long, and in a couple cases even much longer than the matrix plastic by itself. Thirdly, there was no reliably discernable difference in results obtained with very small cellulosic particles in comparison to using very large ones. Although the most common observation was for reinforcement to greatly decrease the elongation to break when compared to the corresponding neat HDPE, there was a lot of variation evident among different studies.



Fig. 4. Published results for the change in tensile modulus, relative to that of the HDPE matrix employed in each study, *vs.* the length (usually determined by screening) of a cellulose-based reinforcement

Flexural strength

Figure 5 shows data for flexural strength, *i.e.* the amount of bending force needed to cause breakage of the HDPE composites. Notably, for this analysis there were no data corresponding to the smallest of the particles, the cellulose nanocrystals (CNC). Although the range of particle size was more limited than in the case of tensile strength testing considered above, the data from the studies reveal a trend toward increasing flexural strength with increasing size of the reinforcement, especially when considering systems with compatibilizing treatments (plotted square symbols). When considering all the data, the R^2 value was 0.203 and the 95% confidence interval for the slope of the linear

regression line was from 0.226 to 0.820, indicating a statistically significant increase in flexural strength with increasing particle size. When considering just the systems with compatibilization (24 data points), the R^2 value was 0.397 and the 95% confidence range for the slope was 0.317 to 1.078, again indicating a statistically significant trend of increasing flexural strength with increasing particle size.



Fig. 5. Published results for the change in flexural strength, relative to that of the HDPE matrix employed in each study, *vs.* the length (usually determined by screening) of a cellulose-based reinforcement



Fig. 6. Published results for the change in flexural modulus, relative to that of the HDPE matrix employed in each study, *vs.* the length (usually determined by screening) of a cellulose-based reinforcement

Flexural modulus

As shown in Fig. 6, a positive trend vs. the logarithm of particle size also was apparent in the plotted results for flexural modulus. When considering all the data together, the R^2 value was 0.216 and the 95% confidence range for the slope was 0.130 to 0.519, indicating a significant positive trend of flexural modulus vs. particle size. When considering only the systems with compatibilization, the R^2 value was 0.294 and the 95% confidence range for the slope was 0.294 and the 95% confidence range for the slope was 0.080 to 0.544, again indicating a significant positive trend in flexural modulus vs. the logarithm of particle size of the reinforcements.

Studies of HDPE in which particle size was an independent parameter

While most of the published studies involving HDPE composites with cellulosebased reinforcements were focused on other composite preparation issues, some studies compared results for particles having different size under matched processing conditions. Because some of these articles did not include evaluation of the strength of the neat HDPE by itself, the data are plotted as actual rather than relative values. Figure 7 shows the data for tensile breaking stress (MPa). Note that the lines shown in the figure connect data points corresponding to the same published article, but with differing sizes of reinforcing particles.



Fig. 7. Published results for tensile breaking strength of HDPE composites for studies that considered more than one particle size under the same conditions of composite preparation. Articles represented in the figure are as follows: (Raj and Kokta 1991; Bouafif *et al.* 2010; Aji *et al.* 2011; Petchwattana *et al.* 2012; Gallagher and McDonald 2013; Pollanen *et al.* 2013; Li *et al.* 2014; Ogah and Afiukwa 2014; Gozdecki and Wilczynski 2015; Tisserat *et al.* 2015; Xu *et al.* 2016; Chen *et al.* 2018; Igarishi *et al.* 2018).

Two main findings can be concluded from a general inspection of the results presented in Fig. 7. First, any effects that might be attributed to particle size, within given studies, were generally small to moderate in terms of the relative difference in tensile strength. Second, while there were a few studies that showed instances of decreasing composite strength with increasing particle size, most of the studies showed the contrary. Thus, the results support the idea that larger particle size is usually associated with moderately higher tensile breaking stress.

A decrease in tensile strength with increasing particle size is shown by the data of Igarishi *et al.* (2018). These researchers compared nanofibrillated cellulose versus bleached softwood kraft fibers. The higher strength achieved with the NFC possibly might be explained by an expected higher aspect ratio, leading to an enhanced reinforcing ability.

Raj and Kokta (1991) observed decreased composite strength with increasing size of reinforcements obtained from the same aspen wood chips. Chemithermomechanical pulp (CTMP) was made using a pressurized refining process, which is a commonly used process in the pulp and paper industry. Wood flour was obtained by grinding the same chips in a mill. The authors did not offer any explanation for the somewhat different performance of the two types of reinforcement.

Tisserat *et al.* (2015a) explained their observed trend of decreasing tensile strength with increasing particle size by reference to particle shape. They observed that during the milling of paulownia wood, the smaller particles tended to have a higher aspect ratio, enhancing their reinforcing capability. Note, however, that the three smallest plotted categories from that study gave approximately the same strength, representing 1.24 to 1.30 times the tensile strength of the neat HDPE.

Figure 8 shows a plot for the tensile modulus, from studies evaluating reinforcing particles of different sizes under the same conditions of HDPE composite preparation. Here the general trend of almost every study was an increased tensile modulus with increasing size of the cellulose-based reinforcement particles.



Fig. 8. Published results for tensile modulus of HDPE composites for studies that considered more than one particle size under the same conditions of composite preparation. Articles represented in the figure are as follows: (Aji *et al.* 2011; Bouafif *et al.* 2010; Chen *et al.* 2018; Gallagher and McDonald 2013; Gozdecki and Wilczynski 2015; Li *et al.* 2014; Pollanen *et al.* 2013; Raj and Kokta 1991; Tisserat *et al.* 2015).

It is apparent that the magnitudes of tensile modulus achieved showed large differences from study to study. It should be borne in mind that the attributes of the base HDPE can be expected to vary from study to study, including the molecular mass, polymer dispersity, crystallinity, and even the purity. Some of the studies employed recycled HDPE, but notably the two studies reporting the lowest tensile modulus results (Aji *et al.* 2011; Tisserat *et al.* 2015b) did not indicate usage of recycled HDPE.

General findings for HDPE

Based on results from the studies dealing with HDPE as a matrix (Figs. 2 to 8), some general summaries can be drawn:

- Measures to improve the compatibility between the HDPE and the cellulose-based reinforcement tend to have a major effect, which is typically more important than other factors such as particle size. To achieve this, some of the most-reported systems for compatibilization include surface modification of the cellulosic material with trimethoxy silane derivatives (Ben Mbarek *et al.* 2013; Chen *et al.* 2017; Fang *et al.* 2017; Zhou *et al.* 2017; Gao *et al.* 2019) or esterification (Zhang *et al.* 2009; Taib *et al.* 2010; Cetin *et al.* 2015). The most popular coupling agents that are typically added to the matrix material before compounding or compression molding (hot-press) are maleic anhydride polyethylene (MAPE) (Taib *et al.* 2010; Li *et al.* 2012; Gallagher and McDonald 2013; Liu *et al.* 2013; Tisserat *et al.* 2013; Migneault *et al.* 2015; Roumeli *et al.* 2015; Chang *et al.* 2016; Cisneros-Lopez *et al.* 2017; Huang *et al.* 2018; Zhang *et al.* 2018; Ihamounchen *et al.* 2019) and maleic anhydride polypropylene (MAPP) (Cui *et al.* 2008; John *et al.* 2010; Elloumi *et al.* 2016).
- For tensile breaking strength, considering many studies for which the range of particle sizes spanned five orders of magnitude, there was generally only a weak positive relationship, if any, between mechanical strength and particle size of the reinforcements. This relationship was not statistically significant when considering all the data together. However, the relationship was demonstrated in several studies where particles of different sizes were compared in HDPE under matched conditions (Bouafif *et al.* 2010; Petchwattana *et al.* 2012; Gallagher and McDonald 2013; Pollanen *et al.* 2013; Li *et al.* 2014; Gozdecki and Wilcynski 2015; Xu *et al.* 2016).
- For tensile modulus and flexural modulus, there was a statistically significant increase in modulus with increasing size of the reinforcing particles, especially in the case of systems with some form of compatibilization, meaning a surface treatment or use of a coupling agent added to the matrix material before compounding. This finding is tentatively attributed to the fact that tests of flexural modulus are likely to involve stress levels below what will cause interfacial separation. Because of this, the greater stiffness of cellulosic particles, relative to typical HDPE, are able to make a significant contribution to composite stiffness.
- Evaluation of test data for elongation to failure gave diverse results, but the majority of the findings showed greatly reduced elongation in the presence of reinforcing particles, whether these were large or very small. These findings are consistent with separation occurring at interfaces, leading to brittle failure and the propagation of cracks at weak points.

The numerical results of statistical analysis, as reported in this section, showed several cases in which either there was a lack of any significant relationship or where higher particle size gave a statistically significant advantage in terms of strength. There are reasons to be cautious regarding those cases that appeared to show statistically significance. First, it can be argued that the data included in such analysis does not truly meet the requirement of independence. Rather, data obtained from different studies can be regarded as showing systematic rather than random differences. Second, it can be argued that the combined data set obtained from different studies would not necessarily involve a normal distribution of values.

When considered as a whole, the take-away message from Part A can be stated as follows: For the first time, the relative properties have been compared for a very large number of composites prepared with cellulose-based reinforcing particles having very different sizes, all in the same nominal matrix material (HDPE). Statistical analysis of the relationships between properties and particle size showed cases of either no significant relationship or a weak but significant positive relationship. In no case did the regression analysis support any inherent advantage of small particles when the goal is to achieve higher composite strength properties.

Part B: Studies Comparing Differently Sized Reinforcements

Having considered a broad range of data, in the case of HDPE, the rest of this review will focus on studies in which particles of different size were compared under matched conditions as reinforcements in matrix polymers other than HDPE. Due to differences in properties such as elastic modulus, surface energy, and hydrogen bonding ability, it should not be necessarily assumed that effects of cellulosic reinforcement particles would follow the same trends as have been discussed for HPDE. An emphasis here will be on studies in which the researchers tested cellulose-based particles of different size in a common matrix material.

Polypropylene

Studies in which polypropylene served as the matrix, and cellulose particles of different size were considered as reinforcements are listed in Table 4:

Table 4. Studies Considering Effects of Particle Size of Cellulose-based

 Reinforcement in Polypropylene

System Considered	Reference
Pine particles of 10 mesh and 50 mesh compared	Wechsler and Hiziroglu 2007
Oak and pine wood fibers; effects of extraction	Ashori & Nourbakhsh 2010
Nut shells and argan particles	Essabir <i>et al.</i> 2013
Wood: four length distributions, three damage levels	Thumm & Dickson 2013
Breakdown of cellulose particles during compounding	Soccalingame et al. 2015
Wood fibers; emphasis on rheological behavior	Durmus et al. 2019
Nanofibers and nanocrystals were compared	Neves et al. 2019
Henequen strands; optimization of coupling agent dose	Tarrés et al. 2019

Ashori *et al.* (2010) observed generally higher breaking stress and modulus values (both tensile and flexural) for polypropylene wood fiber composites reinforced with pine fibers, compared to oak fibers, which are smaller. In contrast, Durmus *et al.* (2019) did not

observe a consistent trend with increasing size classification of the wood fibers; rather, they obtained higher tensile strength and modulus for both larger (350 to 425 μ m) and smaller (< 250 μ m) reinforcements in comparison to intermediate size (250 to 350 μ m). Thumm and Dickson (2013) found that the length of wood fiber used for reinforcement of polypropylene did not have a large effect unless it fell below a critical value of about 0.8 mm. By contrast, fiber distributions having a mean length of about 1.3 mm achieved equally good flexural strength and modulus as longer fibers (*e.g.* 2 mm and 3 mm average fiber length). When comparing nanofibrillated cellulose (NFC) with cellulose nanocrystals (CNC), Neves *et al.* (2019) found substantial increases in compressive strength of the resulting polypropylene composite foams only in the case of the NFC particles, which were larger. Wechsler and Hiziroglu (2007) observed no significant difference in modulus of elasticity when comparing polypropylene composites prepared with pine particles of different size.

Essabir *et al.* (2013) reported an increase in elastic modulus with a decrease in average particle size. All of their composites showed lower tensile breaking strength in comparison with the neat polypropylene. These results are consistent with poor adhesion between the matrix and reinforcement. Since relatively large reinforcing particles, when poorly adhering, can be expected to provide large defects in the structure, such a mechanism is consistent with the size dependency of the mechanical strength results.

Polyamides

Polyamides, in the family of nylons, are well known as strong thermoplastic polymers. Because they have a more polar nature than the HDPE and polypropylene just considered, they can be expected to have better interfacial adhesion to unmodified cellulosic articles. However, the higher temperature processing requirements of polyamides can restrict the use of wood-based materials. Relatively pure cellulose fibers or particles, which have a relatively high thermal tolerance (Yang *et al.* 2007), may be required.

Peng *et al.* (2015) carried out an especially interesting study in which polyamide resin was compounded with three contrasting sizes of cellulose particles: cellulose nanocrystals (CMC), nanofibrillated cellulose (NFC), and microcrystalline cellulose (MCC). Due to air-drying of these particle forms, some inherent clustering of the particles was expected. Detectable gains in tensile modulus were observed with all three types of reinforcement, but the most favorable choice of reinforcement was dependent on the level of loading. At a loading level of 2.5%, the CNC gave the largest increase in tensile modulus (a factor of about 1.08 relative to the neat polyamide). At an intermediate addition level of 5% the NFC gave the biggest reinforcing effect (factor = 1.19), and at the highest addition level of 10% the MCC gave the overall greatest increase in tensile modulus (factor = 1.30). Since the larger MCC particles have lower surface area per unit mass, such results suggest that there may be an optimum amount of surface area of reinforcement in the studied system.

Considering much larger particle sizes, Nishitani *et al.* (2016) filled polyamide matrices with hemp fibers that had been previously cut to lengths of 5, 10, 20, and 50 mm. The material was surface modified using a two-fold treatment with alkaline extraction and silanization treatment to improve interfacial adhesion. Following extrusion, the tensile strength and modulus results both increased with an increase of initial particle size from 5 to 10 mm. Further increases in length yielded lesser benefits, which suggested to the

researchers that the longer fibers were at least partly shortened during the process of extrusion.

Poly-(lactic acid) (PLA)

Probably because PLA is a thermoplastic polymer that can be readily obtained by processing of chemicals derived from plant sources, there has been intense interest in exploring its potential uses and modifications. PLA is a polyester, and it can be either extruded or cast from solvent. Due to the weak, relatively brittle nature of neat PLA, it is widely recognized that it needs to be modified, for instance by use of plasticizers and reinforcing particles (Farah *et al.* 2016). Selected articles in which cellulose-based particles were used to reinforce PLA composites are listed in Table 5.

Table 5.	Studies in Which	Cellulosic	Particles	Were	Used as	Reinforce	ment in
PLA Com	nposites						

System Considered	Reference
CNC compared with MFC fibers; silane treatment	Frone <i>et al.</i> 2011
NFC compared with ordinary wood-pulp fibers	Kowalczyk et al. 2011
Ball-milled celluloses; extrusion; fractal analysis	Gao & Qiang 2017
Bamboo fiber of different lengths	Jiang <i>et al.</i> 2018
Ball-milling for different times; effects of mean size	Qiang <i>et al.</i> 2018
Regenerated cellulose microfibrils compared with NFC	Tanpichai & W. 2018
Cotton fibers with starch as compatibilizer	de Macedo et al. 2019
NFC used with extrusion and injection molding	Peric <i>et al.</i> 2019
CNC, with attention to problems of poor distribution	Wang <i>et al.</i> 2019b

Several of the listed studies involved direct comparison of different sizes of cellulose reinforcement particles under matched conditions of PLA composite preparation. As shown in Fig. 9, Frone *et al.* showed that neither MCC nor MFC, either with or without silane treatment of the cellulosic surfaces, was able to increase the elastic modulus relative to the neat PLA polymer.



Fig. 9. Effects microcrystalline cellulose (MCC) and microfibers on the storage modulus of PLA (data of Frone *et al.* 2011, plotted for the first time)

The silane treatment was shown to be very important, making it possible (especially in the case of the MCC), to essentially match the modulus obtained with neat PLA. In contrast, a thermal annealing treatment yielded substantial increases in modulus in all cases, with the best overall results for the unfilled PLA.

Jiang *et al.* (2018) compared the effects of bamboo-derived fibers of different length classes (<1 mm; 4 to 5 mm; 9 to 10 mm). Tensile and flexural strength and modulus were all increased with the addition of reinforcing fibers of any of the size ranges. The highest mechanical properties were generally achieved with the intermediate size of particles. However, the highest tensile strength was obtained with the longest bamboo fibers. Microscopic evidence indicated good adhesion between the fibers and the PLA matrix in all cases. The authors attributed the poorer performance of the longest fibers, in most cases, to more extensive fiber entanglement, which was confirmed by microscopic images.

Kowalczyk *et al.* (2011) compared the effects of regenerated cellulose fibers of two highly contrasting sizes. SEM micrographs showed that a class called "CSFs" had diameters of 10 to 70 μ m, whereas fibers called "CNFs" had diameters of 200 to 300 nm. Based on mechanical tests at room temperature, the smaller (CNF) particles gave higher tensile and flexural strength at a 2% level of loading. However, the highest tensile strength was found at a 20% loading when using the larger (CSF) reinforcement.

In their study of ball-milled bleached softwood kraft pulp, Qiang *et al.* (2018) found mixed results for the effects of particle size in PLA composites. In the case of tensile strength, the reinforced PLA always gave lower stress to breakage in comparison to the neat PLA. However, the most favorable results, for maintaining tensile strength, were at the highest duration of ball milling, consistent with the smallest reinforcing particles. For tensile modulus, the presence of reinforcements generally yielded higher values, especially at the highest level (20 wt%) of loading. The cellulose that had been ball-milled for the least time yielded the greatest increase in modulus. Overall, the results are consistent with the idea that the evaluation of modulus generally does not involve stresses large enough to cause interfacial separation, thereby allowing the longest fibers to more fully express their stiffening effect. By contrast, the longer fibers would be expected to provide large areas of weakness and likely points of crack initiation during tensile strength testing.

In another study comparing effects of regenerated cellulose fibers with nanofibrillated fibers prepared therefrom, Tanpichai and Wootthikanokkhan (2018) reported higher tensile breaking strength and modulus in the case of the smaller particles, the NFC. The authors attributed the better results, at least in part, to a better interaction of the NFC with the matrix.

Other Polyesters

Rahman *et al.* (2018) studied flat-pressed composites of recycled polyester (PET) with wood particles of different size. Relatively large wood particles at a loading level of 50% gave the highest values of modulus of elasticity. The composite product also had lower density than the base polymer. Though the cited authors did not employ a coupling agent in their work, the results suggest that a coupling agent could have enabled higher strength properties.

Butron *et al.* (2019) recently used a solvent-casting method to prepare films based on a bio-based polyester with cellulose nanocrystals. The solvent casting method permits high levels of reinforcement without damaging the cellulosic material or reducing the particle length. The elastic modulus of the resulting films increased almost linearly with CNC content up to 50%, the highest loading considered. However, the composites with such high amounts of CNC were described as very brittle; the maximum stress was about half that of the base polymer. At a 20% loading of CNC the stress to breakage was equal to that of the neat polyester. The highest stresses and strains to breakage were obtained with loadings in the range 2 to 5% by weight, which is a common level when compounding with CNC.

Polyurethanes

Tian *et al.* (2019) noted that hydrogen bonding can be expected to play an important role in the interaction between cellulose and polyurethane. They used this concept in explaining their promising results with cellulose nanocrystals as a filler for polyurethane. Substantial increases in both tensile modulus and strength to breakage were observed.

Chen *et al.* (2019) prepared composites from waterborne polyurethane filled with nanofibrillated cellulose at a loading of 3% for use in 3D printing. The NFC particles could be added either during polymer synthesis or to the emulsion after its preparation. The casting method of preparation allowed film preparation with little chance of damage to the reinforcing particles. Large increases in elastic modulus were achieved, while maintaining the stress to breakage in systems prepared with an excess amount of trimethylamine (TEA), which was found to increase the viscosity of the resulting formulations.

None of the reports involving polyurethane composites compared cellulose-based particles of differing size. However, cited work (Chen *et al.* 2019, Tian *et al.* 2019) suggests that this would be a good candidate for studies with different sizes of cellulose reinforcement. That is because certain polyester resins can be cured with low to moderate temperature, avoiding damage to cellulosic reinforcements.

Poly-(ethylene oxide) and related

Xu *et al.* (2013) compared polyethylene oxide composites filled with either cellulose nanocrystals or nanofibrils. The larger NFC particles conferred higher strength and modulus to cast films, which was attributed to their higher aspect ratio. Sapkota *et al.* (2017) similarly compared two different CNC types having different aspect ratio; their matrix polymer was ethylene oxide-epichlorohydrin. The results were strongly affected by the presence of some particles having higher aspect ratio. Favorable results, in terms of composite stiffness, were achieved with 50 to 75% of the longer CNCs.

Acrylics

As listed in Table 6, several research groups have obtained promising results for various acrylic polymers filled with nanocellulose particles. These systems offer greater flexibility for composite preparation.

Table 6. Studies of Acrylic Polymers Filled with Nanocellulose

System	Reference
Polybutyl methacrylate with CNC; casting method	Ding et al. 2019
Bacterial cellulose; <i>in-situ</i> polymerization in the mixture	Faria <i>et al.</i> 2019
Copolymer with CNC and <i>in-situ</i> polymerization	Ouzas et al. 2019
Polyacrylamide with carboxymethylated NFC; film casting	Ryu <i>et al.</i> 2019
PMMA-grafted CNCs in acrylate copolymer; film casting	Zhang <i>et al.</i> 2019a,b

Since polymerization can be induced *in-situ* in a mixture by use of a free radical catalyst, the composite can be prepared from a solution, or from a melt. Although these listed studies show a potential role for very small cellulose-based particles, none of them compared particles of different size under the same conditions.

Epoxy resin systems

As epoxy resins can be prepared from a low-viscosity mixture, as well as having strong adhesion to polar surfaces, they can be regarded as ideal for many applications involving cellulosic reinforcements.

Effects of particle size were considered by He *et al.* (2016), who studied the reinforcement of an epoxy matrix with poplar wood flour and polyamine. With decreasing size of the wood flour particles, the mechanical properties initially increased but then showed a minor downward trend. The mechanisms of failure, based on microscopic observations, included both fiber breakage and fiber pullout, the latter indicating interfacial separation. The authors also reported evidence of agglomeration of the reinforcing material, especially at higher levels of addition. Longer fibers of similar width are more prone to agglomeration during stirring of a mixture (Hubbe 2017), and such agglomerates can serve as weak points in a composite. Thus, more extensive agglomeration is a possible reason to explain the minor fall-off in strength properties with increasing size of wood flour particles. Notably, Barczewski *et al.* (2019), who worked with ground sunflower husk and nut shells, found better strength performance with low-aspect-ratio particles. This was attributed to the fact that fibrous particles, having high aspect ratio, increased the viscosity of the mixture, leading to a porous structure of the composite.

Various other studies with epoxy resins showed promise for the preparation of nanocomposites. Nair *et al.* (2019) obtained promising results with NFC both in the glassy region and in the transition region of the matrix polymer. Saikia *et al.* (2019) also studied effects of NFC and reported typical dimensions of 40 nm thickness and 10 μ m length. Increased mechanical properties were observed, along with transparency to light. Given that epoxy-based composites are relatively easy to prepare with a wide range of reinforcements and a high tolerance for polar surfaces, it is to be hoped that a future study will explore the effects of particle size and length in greater depth.

Phenol formaldehyde (PF) and related resin systems

PF resins, as in the case of epoxies, have the advantage of relatively low viscosity prior to curing, making it possible to prepare composites with a wide range of reinforcements, some of which might be breakable under high shear or intolerant of the temperatures in a polymer melt. With respect to particle size, some of the most interesting results involving PF resin composites were those reported by Neelamana *et al.* (2013). These researchers compared banana fibers (fiber diameter 80 μ m) with microfibers and nanofibers obtained by specific processing steps, including both chemical treatments and steam explosion. As shown in Fig. 10, the nanofibers were found to be about twice as effective as the other classes of particle, meaning that about half the dosage was needed to achieve a given increase in tensile strength.

Figure 11 shows corresponding results for flexural modulus. Here the nanofibers again showed the greatest effectiveness, but there was an interesting differentiation involving the two larger size classes of reinforcement. Notably, the relatively large unmodified banana fibers outperformed the intermediate class, the microfibers. The authors did not suggest any explanation. However, it seems likely that the complex, partly

delaminated structure of woody material subjected to an initial steam explosion treatment would be susceptible to internal breakage, thus providing weak points at which cracks can be initiated.



Fig. 10. Effects of three contrasting cellulosic particle reinforcements on the tensile strength of phenol-formaldehyde composites. Fibers = banana pseudo-stem; microfibers were obtained after soaking banana fibers in 2% NaOH, steam explosion, bleaching, washing, and drying; nanofibers were obtained from the microfibers with further eight-time steam explosion in the presence of oxalic acid. Figure plotted for the first time from the reported data (Neelamana *et al.* 2013)



Fig. 11. Effects of three contrasting cellulosic particle reinforcements on the flexural strength of phenol-formaldehyde composites. Figure plotted for the first time from the reported data (Neelamana *et al.* 2013)

Poly-(vinyl alcohol) (PVOH)

Due to the generally hydrophilic nature of cellulose-based fibers, it is important to consider water-soluble polymers as a potential matrix. Among the synthetic water-soluble polymers, PVOH has been the most often studied for such purposes. Effects of particle

size were considered by Jiang *et al.* (2018). Microfibrillated cellulose, having different attributes, was obtained from pure cellulose, holocellulose, and kraft pulp. For all three types of starting material, the tensile strength and elastic modulus both increased with decreasing size of the MFC. It was concluded that subjecting the cellulose to more prolonged wet-milling conditions improved its performance in the PVOH composites.

Starch and other biopolymers

Different varieties of starches are widely used in adhesives, and they tend to be cost-effective, particularly in cases where the resulting properties are sufficient to meet the application requirements. This includes many paper industry applications.

Soykeabkaew *et al.* (2012) compared the effects of bacterial cellulose (nano-sized) *vs.* ordinary jute fibers in starch composites prepared by film casting from aqueous solution. The highest strength results were achieved at a remarkably high loading of 50% bacterial cellulose. Relative to the strength of the starch by itself, the tensile strength was increased by a factor of about 200, and the tensile modulus was increased by an even greater factor. In general terms, these results are consistent with a very strong adhesion between the cellulose and the starch matrix, the weakness of the starch by itself, and the very high aspect ratio of the bacterial cellulose nanoparticles.

Ali *et al.* (2017) reported the effects of particle size of wheat and corn hull material in starch-based composite films. Only very weak dependencies on particle size were observed.

Summary for studies with different sizes of particles in specified matrices

Although studies comparing different sizes of cellulose-based particles under matched conditions in selected matrix systems yielded diverse results, some general trends can be identified from those cases discussed:

- As may be anticipated, experimental findings generally indicate that greater relative gains in elastic modulus can be achieved with softer matrix materials, especially starch, but also including HDPE.
- Advantages associated with longer fibers tended to be most pronounced relative to tensile and flexural modulus values. These advantages for longer fibers tended to be weaker or not present when considering tensile or flexural breaking strength testing.
- Compared to other forms of cellulose-based reinforcements, nanofibrillated cellulose often stands out as giving a superior contribution to strength of composites. This might be related to an expected high aspect ratio, typical of NFC.
- Relatively polar and in particular hydrogen-bonding matrix materials appear to offer a key advantage for cellulose-based reinforcements, since strong adhesion generally can be achieved in such cases without the need for any surface treatment of the reinforcement particles or addition of any coupling agent to the matrix.

Disclaimers

Before ending this review, it seems appropriate to draw attention to certain issues that might be considered in follow-up investigations. One issue, which was only dealt with to a minor degree in the present article, is the effects of different processing methods for composites. Another key issue concerns possible correlations involving the aspect ratios or diameters of reinforcing particles.

Effects of processing options

The properties of the neat HDPE employed in different studies showed very wide variations. It is not possible to know, when comparing results from such studies, whether the differences in their findings can be attributed to differences in the source materials employed or to differences in processing conditions, some as simple as the cooling rates The processing conditions are of special concern when used in injection molding. attempting to gain information about particle size effects from studies involving twin-screw extrusion, since the shear forces have been shown to be sufficient in many cases to shorten the fibers (Dickson et al. 2014; Soccalingame et al. 2015; Dickson and Sandquist 2018). In future research there are several potential strategies that could be considered in dealing with this issue. Perhaps the simplest is just to use relatively low-aspect-ratio reinforcing particles in such cases (Barczewski et al. 2019). Another approach is to essentially allow the high shear conditions associated with compounding and extrusion processes to do the work of size reduction and shortening. In such cases, if there is interest in investigating the effects of the length of reinforcing particles, the fiber length data needed for analysis could be obtained by microscopic imaging or by dissolving the matrix polymer, thus making it possible to recover the reinforcing particles and evaluate their size distributions. A third approach, in cases where relatively large gains in elastic modulus properties are sought, may be to employ manufacturing methods other than extrusion, *e.g.* solvent casting.

Other sources of variability

Another point to keep in mind, while considering the comparisons and regressions that have been made in the course of this article, is that there can be very large differences in results for nominally similar composite materials. Potential explanations for wide ranges of reported findings, when reporting results for the same matrix and reinforcement, can include differences in the matrix (including the modulus, degree of crystallinity, molecular mass, and impurities, *etc.*) and the reinforcing particles. In the latter case there may be unreported and hard-to-quantify differences in fiber damage, including curl and kink. In addition, the resulting orientation, as well as the extent of dispersion or flocculation in the composite, is likely to vary a lot, even within a given study. An analysis of the effects of such parameters on composite attributes would best be carried out in a dedicated experiment with well-characterized matrix materials and with fibers having relatively simple shapes.

Selection of Particle Sizes for Different Composite Applications

Though the regression analyses carried out in preparation of this article did not find any strong relationships between the particle size of reinforcing elements on composite properties, there still can be other reasons to strongly favor different ranges of size in different applications. Very small particles, including nanomaterials, may be strongly favored in applications where the material must flow through narrow orifices, such as nozzles for three-dimensional printing (Xu *et al.* 2015). Also, smaller reinforcements are desirable in cases where very smooth surfaces of plastic composites are a priority (Wechsler and Hiziroglu 2007). On the other hand, when producing large-scale materials at relatively low cost, the use of relative large cellulose-based reinforcements has the potential to save in preparation costs of the particles themselves, in addition to reducing any costs associated with compatibilizers to improve adhesion with the matrix (George *et al.* 2001; Belgacem and Gandini 2005; Kumar *et al.* 2011).

CONCLUDING REMARKS

As was noted in the Introduction, in preparing this review article the authors have attempted to determine whether there is an advantage of using large or small cellulose-based particles for reinforcement of various composites. In particular, the goal was to examine published evidence related to a possible strength advantage when nanocellulose reinforcements are used (Fu *et al.* 2008; Tanpichai and Wootthikanokkhan 2018). The overall findings from this review generally do not support the presumption that nanocellulose reinforcements offer a consistent and dependable advantage in terms of composite strength or modulus.

Having stated the general findings drawn from a "big picture" view of results from numerous studies, it is also possible to draw some other lessons from the research that was considered in preparing this article.

- For the reinforcement of low-surface-energy plastics such as HDPE and polypropylene, where it has been shown that either surface treatment of the cellulosic particles (*e.g.* silane treatment or esterification) or addition of a coupling agent to the plastic before compounding and extrusion (*e.g.* MAPE), the weight of the evidence favors the use of relatively large particles, within the range of practicality. The upper end of that practical range might be related to effects on surface roughness or appearance of a composite, process limitations due to the sizes of nozzles, *etc.*, and the possibility that too-large cellulosic particles might function as sites of initiation of cracks that are big enough to precipitate early failure in tensile tests or flexural strength tests.
- For the reinforcement of polar matrix materials such as epoxy resin, starch, PLA, *etc.*, which bond relatively strongly to unmodified cellulosic surfaces, the motivation to favor larger particle sizes for reinforcements seems less strong. Thus, it is easier to justify serious consideration of a range of different particle sizes, depending on the details of the property objectives for the composites.
- Both economics and sustainability principles generally favor usage of plant-based materials with the least amount of modification that is required to meet one's objectives. Such principles generally favor using cellulosic material in the largest dimensions consistent with both the processing limitations and the required product attributes.

ACKNOWLEDGEMENTS

The authors thank the following individuals who studied an earlier draft and provided suggestions and corrections: Parameswaranpillai Jyotishkumar, Research Professor, King Mongkut's University of Technology North Bangkok (KMUTNB), 1518 Pracharat 1, Wongsawang Road, Bangsue, Bangkok, 10800, Thailand; David Harper, Univ Tennessee, Ctr Renewable Carbon, 2506 Jacob Dr, Knoxville, TN 37996 USA; and Salim

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APPENDIX

Calculations to Estimate Specific Surface Area of Cellulose Nanocrystals (CNCs) and Softwood Kraft Fibers

Cellulose nanocrystals (CNC)

Assume the dimensions shown in the following figure, and a density of 1.5 g/cm³.



Area of one crystal $\approx 5 \text{ nm X } 100 \text{ nm X } 4 = 2000 \text{ nm}^2 = 2 \text{ X } 10^{-15} \text{ m}^2$ Volume of one crystal = 5 nm X 5 nm X 100 nm = 2500 nm³ = 2.5 X 10⁻¹⁸ cm³ Mass of one crystal = 1.5 g/cm³ X 2.4 X 10⁻¹⁸ cm³ = 3.75 x 10⁻¹⁸ g Area per unit mass $\approx 2 \text{ X } 10^{-15} \text{ m}^2 / 3.75 \text{ x } 10^{-18} \text{ g} = 533 \text{ m}^2/\text{g}$

Softwood fibers

Assume the dimensions shown in the following figure, and a density of 0.5 g/cm³.



Area of one fiber $\approx 50 \ \mu m \ X \ 2000 \ \mu m \ X \ 4 = 4 \ X \ 10^5 \ \mu m^2 = 4 \ X \ 10^{-7} \ m^2$ Volume of one fiber $\approx 50 \ \mu m \ X \ 50 \ \mu m \ X \ 2000 \ \mu m = 5 \ X \ 10^6 \ \mu m^3 = 5 \ X \ 10^{-6} \ cm^3$ Mass of one fiber $\approx 0.5 \ g/cm^3 \ X \ 5 \ X \ 10^{-6} \ cm^3 = 2.5 \ x \ 10^{-6} \ g$ Area per unit mass $\approx 4 \ X \ 10^{-7} \ m^2 / \ 2.5 \ x \ 10^{-6} \ g \approx 1 \ m^2/g$

Matrix type	Com- Pat?	Reinforcement			Tensile Properties			Flexural Properties		Reference	
		Туре	Level	Length	Log	T. Str.	T. Mod.	Elong	Fl. Str.	Fl. Mod.	
			(%)	(um)	(Length)	(rel.)	(rel.)	(rel)	(MPa)	(MPa)	
HDPE	Yes	Millet husk fiber	10	1000	3.00	1.10	1.40	0.02	1.18	1.11	Abba <i>et al.</i> 2017
HDPE	No	Pine cone powder	10	1000	3.00	1.12			1.26	1.25	Agayev & Ozdemir 2019
HDPE	Yes	Wood	30	250	2.40	1.45	3.27	0.37	1.85		Aggarwal <i>et al.</i> 2013
HDPE	No	Wood	30	250	2.40	1.04	3.18	0.34	1.09		Aggarwal <i>et al.</i> 2013
HDPE	No	Kenaf/PALF	30	250	2.40	0.89	1.26		1.00	1.15	Aji <i>et al.</i> 2011
HDPE	No	Kenaf/PALF	30	500	2.70	0.92	1.39		1.10	1.25	Aji <i>et al.</i> 2011
HDPE	No	Kenaf/PALF	30	750	2.88	0.91	1.58		1.20	1.50	Aji <i>et al.</i> 2011
HDPE	No	Kenaf/PALF	30	2000	3.30	0.92	1.52		1.32	1.75	Aji <i>et al.</i> 2011
HDPE	Yes	Teak sawdust	15	150	2.18	1.00		0.45			Akter <i>et al.</i> 2018
HDPE	No	Teak sawdust	15	150	2.18	0.80		0.35			Akter <i>et al.</i> 2018
HDPE	Yes	Pine flour	10	285	2.45	1.17					Alavi <i>et al.</i> 2013
HDPE	No	Pine flour	10	285	2.45	0.90					Alavi <i>et al.</i> 2013
HDPE	No	Date pits	10	500	2.70	1.05					Alsewailem & B. 2010
HDPE	No	Microfibers	5	634	2.80	0.70	0.50				Ashok <i>et al.</i> 2019
HDPE	No	Microfibers	5	634	2.80	0.64	0.35	1.20			Ashok <i>et al.</i> 2019
HDPE	No	Oak flour	20	325	2.51	1.00	2.00		1.43	1.50	Bajwa <i>et al.</i> 2019
HDPE	No	Three ag wastes	15	50	1.70	0.70	1.11	0.44	0.80	1.24	Barczewski et al. 2019
HDPE	Yes	Pine fibers	30	200	2.30	1.41	3.27				Ben Mbarek et al. 2013
HDPE	No	Pine fibers	30	200	2.30	0.90	2.24				Ben Mbarek et al. 2013
HDPE	Yes	MCC	10	40	1.60	1.02	1.09		1.20	1.14	Boran 2016
HDPE	Yes	Ultrafine cellulose	4	8	0.90	1.06	0.88	0.95	0.98	1.14	Boran <i>et al.</i> 2016
HDPE	No	Ultrafine cellulose	4	8	0.90	0.95	0.96	0.99	0.95	1.20	Boran <i>et al.</i> 2016
HDPE	Yes	Wood	35	1200	3.08	1.22	2.24		1.57	1.82	Bouafif et al. 2010

Table A. Relative Strength (Compared to the Corresponding Neat HDPE)

HDPE	Yes	Wood	35	3400	3.53	1.47	1.99		2.10	2.27	Bouafif et al. 2010
HDPE	Yes	Wood	35	3900	3.59	1.78	2.95		3.00	5.00	Bouafif <i>et al.</i> 2010
HDPE	Yes	Wood	35	4450	3.65	1.66	2.76		2.73	4.18	Bouafif <i>et al.</i> 2010
HDPE	Yes	Agave fiber	20	348	2.54	0.70	1.09	0.05	1.63	1.77	Cisneros-L. et al. 2017
HDPE	No	Agave fiber	20	348	2.54	1.17	1.22	0.05	1.29	1.65	Cisneros-L. et al. 2017
HDPE	Yes	Deinking sludge	20	264	2.42	0.95	1.47	0.01			Elloumi <i>et al.</i> 2016
HDPE	No	Deinking sludge	20	264	2.42	0.92	1.35	0.02			Elloumi <i>et al.</i> 2016
HDPE	Yes	Hardwood	25	584	2.77	2.35	3.05				Fonseca-V et al. 2015
HDPE	No	Hardwood	25	584	2.77	2.20	2.95				Fonseca-V et al. 2015
HDPE	Yes	Cashew nut shell	30	950	2.98	0.27	0.30	2.25			Gomes <i>et al.</i> 2018
HDPE	Yes	Pine flour	14	189.5	2.28	0.99	1.39				Grison <i>et al.</i> 2016
HDPE	Yes	Pine flour	33	189.5	2.28	1.82	2.03				Grison <i>et al.</i> 2016
HDPE	Yes	Pine	40	200	2.30	1.48	1.78		1.92	1.72	Hillig <i>et al.</i> 2017
HDPE	Yes	Pine + Cocos	40	200	2.30	1.02	1.81		1.24	0.96	Hillig et al. 2017
HDPE	Yes	HW sawdust	20	1000	3.00	1.47			1.38		Hillig <i>et al.</i> 2011
HDPE	Yes	Bagasse powder	40	350	2.54	1.14	1.83		1.44	2.00	Hosseini <i>et al.</i> 2017
HDPE	Yes	Bagasse pulp	40	350	2.54	1.95	3.40		2.28	3.70	Hosseini <i>et al.</i> 2017
HDPE	Yes	NFC SW kraft	10	25	1.40	2.35	3.13	0.03			Igarishi <i>et al.</i> 2018
HDPE	Yes	SW kraft bleached	10	1000	3.00	2.43	3.20	0.03			Igarishi <i>et al.</i> 2018
HDPE	No	SW kraft bleached	10	1000	3.00	1.65	1.99	0.06			Igarishi <i>et al.</i> 2018
HDPE	No	Rubber tree seed	30	100	2.00	0.21	2.73	0.17			Ismail & Shafiq 2016
HDPE	No	Willow fibers	40	400	2.60	0.74			1.03		Jia <i>et al.</i> 2018
HDPE	Yes-hi	Pine milled	40	400	2.60				1.94	3.18	Kajaks <i>et al.</i> 2015
HDPE	Yes-m	Pine milled	40	400	2.60				1.34	2.43	Kajaks <i>et al.</i> 2015
HDPE	No	Birch plywood res.	40	225	2.35	0.64	2.40		1.50	2.40	Kajaks <i>et al.</i> 2018
HDPE	Yes	Sawdust	40	900	2.95	1.22					Khamedi <i>et al.</i> 2019
HDPE	No	Vine stem	10	120	2.08	1.30	1.43		1.73	1.46	Kilinc et al. 2016
HDPE	Yes	NFC/wood	10	15	1.18	1.74	1.18				Li <i>et al.</i> 2016
HDPE	Yes	NFC/wood	10	15	1.18	1.11	1.49				Li <i>et al.</i> 2016

HDPE	Yes	NFC cotton	30	15	1.18	2.57	3.86				Li <i>et al.</i> 2014
HDPE	Yes	NFC cotton	30	15	1.18	1.57	2.27				Li <i>et al.</i> 2014
HDPE	Yes	CNC	20	0.2	-0.70	1.79	1.93				Li <i>et al.</i> 2014
HDPE	No	Peanut shells	10	100	2.00	0.97	1.44				Louvier-H et al. 2018
HDPE	Yes	Eucalypt kraft	30	53.3	1.73	1.55	3.59	0.02			Migneault <i>et al.</i> 2015
HDPE	No	Eucalypt kraft	30	53.3	1.73	1.66	3.33	0.02			Migneault <i>et al.</i> 2015
HDPE	No	CNC	1	0.197	-0.71	1.18	1.42	0.03			Mokhena & Luyt 2014
HDPE	No	CNC	5	0.197	-0.71						Mokhena & Luyt 2014
HDPE	Yes	CNC	1	0.197	-0.71	0.94	1.51	0.33			Mokhena & Luyt 2014
HDPE	Yes	CNC	5	0.197	-0.71	1.92	2.83	0.05			Mokhena & Luyt 2014
HDPE	No	CNC	1	0.197	-0.71	1.47	1.54	0.02			Mokhena & Luyt 2014
HDPE	No	CNC	5	0.197	-0.71		1.56				Mokhena & Luyt 2014
HDPE	Yes	Cane bagasse	20	350	2.54	1.33	1.78	0.73			Mulinari <i>et al</i> . 2010
HDPE	No	Cane bagasse	20	350	2.54	1.01	1.56	0.64			Mulinari <i>et al</i> . 2010
HDPE	No	CNC	1	0.22	-0.66	1.00	0.99	0.96			Novo <i>et al</i> . 2018
HDPE	No	CNC	5	0.22	-0.66	0.73	1.17	0.53			Novo <i>et al.</i> 2018
HDPE	No	CNC	5	0.22	-0.66	0.58	0.91	0.01			Novo <i>et al</i> . 2018
HDPE	No	Corn cob fibers	65	150	2.18				0.64	1.47	Ogah & Afiukwa 2014
HDPE	No	Rice husk fibers	65	125	2.10				0.79	1.65	Ogah & Afiukwa 2014
HDPE	No	Walnut shell	65	165	2.22				0.50	1.12	Ogah & Afiukwa 2014
HDPE	No	Flax shive	65	300	2.48				0.75	1.76	Ogah & Afiukwa 2014
HDPE	Yes	Pine flour	30	340	2.53	1.13	1.75	0.01	1.54	2.00	Ozmen <i>et al</i> . 2013
HDPE	Yes	Pine flour	30	340	2.53	1.23	1.64	0.01	1.75	2.25	Ozmen <i>et al</i> . 2013
HDPE	Yes	Pine flour	30	340	2.53	1.06	1.78	0.01	1.56	2.00	Ozmen <i>et al.</i> 2013
HDPE	Yes	Pine flour	30	340	2.53	0.97	2.00	0.01	1.55	2.03	Ozmen <i>et al</i> . 2013
HDPE	No	Pine flour	30	340	2.53	0.89	1.75	0.01	1.54	2.50	Ozmen <i>et al</i> . 2013
HDPE	Yes	Rice husk	30	75	1.88	1.39			1.50		Petchwattana et al. 2012
HDPE	Yes	Rice husk	30	500	2.70	1.27			1.35		Petchwattana et al. 2012
HDPE	No	Rice husk	30	75	1.88	1.22			1.22		Petchwattana et al. 2012

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HDPE	No	Rice husk	30	500	2.70	1.10			1.18		Petchwattana et al. 2012
HDPE	Yes	Viscose fiber	40	6000	3.78	3.25	5.94	0.02			Pollanen <i>et al.</i> 2013
HDPE	Yes	MCC	40	100	2.00	1.63	3.64	0.03			Pollanen <i>et al</i> . 2013
HDPE	Yes	Aspen CTMP	30	800	2.90	1.11	1.65	0.33			Raj & Kokta 1991
HDPE	No	Aspen CTMP	30	800	2.90	0.93	1.52	0.32			Raj & Kokta 1991
HDPE	Yes	Aspen flour	30	300	2.48	1.38	1.63	0.46			Raj & Kokta 1991
HDPE	No	Aspen flour	30	300	2.48	0.92	1.49	0.37			Raj & Kokta 1991
HDPE	Yes	OCC fibers	30	2000	3.30	1.89	3.50	0.56	1.88	2.06	Rashno <i>et a</i> l. 2018
HDPE	No	OCC fibers	30	2000	3.30	1.32	2.60	0.44	1.40	1.22	Rashno <i>et a</i> l. 2018
HDPE	Yes	Bamboo fiber	30	1147	3.06	1.63	3.00		2.05	3.14	Ren <i>et al.</i> 2014
HDPE	Yes	Hemp	35	2000	3.30	1.11	2.04	0.00			Roumeli <i>et al</i> . 2015
HDPE	No	Hemp	35	2000	3.30	0.69	1.99	0.00			Roumeli <i>et al.</i> 2015
HDPE	Yes	NFC	10	20	1.30	1.84	2.44	0.24			Sakakibara <i>et al</i> . 2016
HDPE	No	NFC	10	20	1.30	1.37	1.81	0.49			Sakakibara <i>et al</i> . 2016
HDPE	No	Wood flour	30	425	2.63		1.96	0.34			Sheng <i>et al</i> . 2017
HDPE	Yes	Teak fiber	15	150	2.18	1.06	1.29	0.08			Sewda & Maiti 2010
HDPE	No	Teak fiber	15	150	2.18	0.88	0.95	0.10			Sewda & Maiti 2010
HDPE	No	Turmeric spent	20	50	1.70	0.85	1.24	0.09			Syed <i>et al</i> . 2010
HDPE	Yes	Acacia fiber	40	800	2.90	0.74	1.73	0.23			Taib <i>et al</i> . 2010
HDPE	No	Acacia fiber	40	800	2.90	0.68	1.84	0.16			Taib <i>et al</i> . 2010
HDPE	Yes	Spruce sawdust	30	500	2.70	1.37	5.00				Tazi <i>et al</i> . 2014
HDPE	No	Spruce sawdust	30	500	2.70	0.74	2.67				Tazi <i>et al</i> . 2014
HDPE	Yes	DDGS	25	300	2.48	0.94	1.08	0.15			Tisserat <i>et al.</i> 2013b
HDPE	No	DDGS	25	300	2.48	0.68	1.05	0.17			Tisserat et al. 2013b
HDPE	Yes	Paulownia wood	25	366	2.56	1.08	1.64	0.06			Tisserat <i>et al</i> . 2015b
HDPE	No	Paulownia wood	25	366	2.56	0.71	1.47	0.07			Tisserat <i>et al.</i> 2015b
HDPE	Yes	Paulownia wood	25	512	2.71	0.85	1.81	0.05			Tisserat <i>et al.</i> 2015b
HDPE	Yes	Paulownia wood	25	215	2.33	1.24	1.71	0.07			Tisserat et al. 2015b
HDPE	Yes	Paulownia wood	25	90	1.95	1.25	1.71	0.07			Tisserat et al. 2015b

HDPE	Yes	Paulownia wood	25	60	1.78	1.26	1.47	0.10			Tisserat <i>et al</i> . 2015b
HDPE	Yes	Rice husk fibers	30	500	2.70		1.45		2.24	1.39	Tong <i>et al.</i> 2014
HDPE	Yes	Douglas fir fiber	50	177	2.25	1.02	3.80				Wang <i>et al</i> . 2019a
HDPE	Yes	Rice husk flour	30	163	2.21		2.24				Yang 2017
HDPE	No	Rice husk flour	30	163	2.21		1.17				Yang 2017
HDPE	Yes	Rice straw	40	1050	3.02	1.38	10.60		1.94	2.10	Yao <i>et al</i> . 2011
HDPE	No	Rice straw	40	1050	3.02	1.07	12.90		1.49	3.40	Yao <i>et al</i> . 2011
HDPE	No	CNF	5	20	1.30	1.17	1.33	0.35	2.25	2.33	Yasim-Anuar <i>et al.</i> 2019
HDPE	Yes	Pine flour	35	250	2.40	1.30					Zabihzadeh <i>et al</i> . 2010
HDPE	No	Pine flour	35	250	2.40	1.08					Zabihzadeh <i>et al.</i> 2010
HDPE	Yes	Poplar flour	35	250	2.40	1.35					Zabihzadeh <i>et al.</i> 2010
HDPE	No	Poplar flour	35	250	2.40	1.16					Zabihzadeh <i>et al</i> . 2010
HDPE	Yes	Pine flour	20	180	2.26				1.04	1.25	Zhang et al. 2009

Table B. Articles Comparing Different Sizes of Reinforcement Particles in HDPE Matrix

Matrix	Com	Reinforcement				Properties					References
Туре	pat?	Туре	Level	Length	Log	T. Str.	T. Mod.		Fl. Str.	Fl. Mod.	
			(%)	(um)	(Len.)	(MPa)	(MPa)		(MPa)	(MPa)	
HDPE	No	Kenaf/PALF	30	250	2.40	28	420				Aji <i>et al.</i> 2011
HDPE	No	Kenaf/PALF	30	500	2.70	30	465				Aji <i>et al.</i> 2011
HDPE	No	Kenaf/PALF	30	750	2.88	29	525				Aji <i>et al.</i> 2011
HDPE	No	Kenaf/PALF	30	2000	3.30	30	510				Aji <i>et al.</i> 2011
HDPE	Yes	Wood	35	1200	3.08	21	1750		23	1040	Bouafif <i>et al.</i> 2010
HDPE	Yes	Wood	35	3400	3.53	26	1550		31	1250	Bouafif <i>et al.</i> 2010
HDPE	Yes	Wood	35	3900	3.59	32	2300		45	2700	Bouafif <i>et al.</i> 2010
HDPE	Yes	Wood	35	4450	3.65	30	2150		41	2300	Bouafif et al. 2010
HDPE	Yes	Corn fibers	50	87.5	1.94	23	2600	1.8	34	2850	Chen <i>et al.</i> 2018
HDPE	Yes	Corn fibers	50	154	2.19	25	2800	1.8	40	3500	Chen <i>et al.</i> 2018
HDPE	Yes	Corn fibers	50	111	2.05	21.5	2650	1.7	35	3150	Chen <i>et al.</i> 2018
HDPE	Yes	Corn fibers	50	199	2.30	26.5	2400	2.3	46.5	3450	Chen <i>et al.</i> 2018
HDPE	No	Wood meal	40	400	2.60				20.7	2218	Chen <i>et al.</i> 2006
HDPE	No	Wood meal	40	665	2.82				14.8	1659	Chen <i>et al.</i> 2006
HDPE	No	Wood meal	40	960	2.98				16.4	1805	Chen <i>et al.</i> 2006
HDPE	No	Wood meal	40	2000	3.30				20.7	2218	Chen <i>et al.</i> 2006
HDPE	Yes	Wood fibers	40	29	1.46	45	1650				Gallagher & M. 2013
HDPE	Yes	Wood fibers	40	109	2.04	46.7	1780				Gallagher & M. 2013
HDPE	Yes	Wood fibers	40	110	2.04	49.2	2010				Gallagher & M. 2013
HDPE	No	Wood fibers	40	29	1.46	35.6	1660				Gallagher & M. 2013
HDPE	No	Wood fibers	40	109	2.04	37.7	1800				Gallagher & M. 2013
HDPE	No	Wood fibers	40	110	2.04	29.7	2020				Gallagher & M. 2013
HDPE	No	SW particles	4	2200	3.34	15	2160		36.8	2640	Gozdecki & W. 2015

HDPE	No	SW particles	40	4600	3.66	16.6	2330		40.6	2790	Gozdecki & W. 2015
HDPE	No	SW particles	40	12700	4.10	18.7	2590		45.8	3150	Gozdecki & W. 2015
HDPE	Yes	NFC SW kraft	10	25	1.40	54		0.03			Igarishi <i>et al.</i> 2018
HDPE	Yes	SW kraft bleach	10	1000	3.00	44		0.027			Igarishi et al. 2018
HDPE	Yes	NFC cotton	30	15	1.18	35	1650				Li <i>et al.</i> 2014
HDPE	Yes	CNC	20	0.2	-0.70	25	850				Li <i>et al</i> . 2014
HDPE	No	Corn cob fibers	65	150	2.18				0.64	1.47	Ogah & Afiukwa 2014
HDPE	No	Rice husk fibers	65	125	2.10				0.79	1.65	Ogah & Afiukwa 2014
HDPE	No	Walnut shell	65	165	2.22				0.5	1.12	Ogah & Afiukwa 2014
HDPE	No	Flax shive	65	300	2.48				0.75	1.76	Ogah & Afiukwa 2014
HDPE	Yes	Rice husk	20	75	1.88	22			23.8		Petchwattana et al. 2012
HDPE	Yes	Rice husk	20	180	2.26	21.5			23.5		Petchwattana et al. 2012
HDPE	Yes	Rice husk	20	250	2.40	21			23.2		Petchwattana et al. 2012
HDPE	Yes	Rice husk	20	355	2.55	20.7			22.9		Petchwattana et al. 2012
HDPE	Yes	Rice husk	20	500	2.70	20			22.9		Petchwattana et al. 2012
HDPE	Yes	Viscose fiber	40	6000	3.78	52.4	3980	0.0175			Pollanen <i>et al</i> . 2013
HDPE	Yes	MCC	40	100	2.00	26.2	2440	0.027			Pollanen <i>et al</i> . 2013
HDPE	Yes	Aspen CTMP	30	800	2.90	27	1800	0.33			Raj & Kokta 1991
HDPE	No	Aspen CTMP	30	800	2.90	23.8	1720	0.32			Raj & Kokta 1991
HDPE	Yes	Aspen flour	30	300	2.48	32	1850	0.46			Raj & Kokta 1991
HDPE	No	Aspen flour	30	300	2.48	23.5	1680	0.37			Raj & Kokta 1991
HDPE	Yes	Paulownia wood	25	60	1.78	20.8	228.1				Tisserat et al. 2015b
HDPE	Yes	Paulownia wood	25	90	1.95	20.6	264.5				Tisserat et al. 2015b
HDPE	Yes	Paulownia wood	25	215	2.33	20.5	265.3				Tisserat et al. 2015b
HDPE	Yes	Paulownia wood	25	366	2.56	16.7	267.7				Tisserat et al. 2015b
HDPE	Yes	Paulownia wood	25	512	2.71	14	280.7				Tisserat et al. 2015b
HDPE	Yes	Rubber seed shell	50	137	2.14	15		5.7	26	3200	Xu <i>et al.</i> 2016
HDPE	Yes	Rubber seed shell	50	163	2.21	16		5	23	3100	Xu <i>et al.</i> 2016
HDPE	Yes	Rubber seed shell	50	210	2.32	16.8		4.6	27.3	3240	Xu <i>et al.</i> 2016