

Flexible and Conductive Cellulose Substrate by Layered Growth of Silver Nanowires and Indium-doped Tin Oxide

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Regenerated cellulose film (RCF) has potential as a conductive substrate due to features such as its degradability, transparency, and flexibility. Indium doped tin oxide (ITO) is a conventional conductive material, but its rigidity restricts the formation of flexible conductive film. In this study, silver nanowires (AgNWs) were introduced between the RCF and the ITO conductive framework. Additionally, the fabrication of flexible, conductive, and transparent RCF was conducted. The AgNWs-ITO based RCF demonstrated high conductivity (170 Ω per sq) and transparency (78%) by the addition of 50 μ L of AgNWs. After bending the sample 50 times with a 5 mm curve radius, the as-prepared conductive RCF presented an electric resistance improvement of 19%, with a 485% increase for the control ITO-based RCF. This is a result of the AgNWs framework, which can lessen the destruction of the bending treatment on the conductive layer and can also desirably connect the ITO conductive sections. The novel approach can expedite the versatile applications of flexibly conductive RCF on printable, portable, and wearable electronic devices.

Keywords: Conductive regenerated cellulose film; Indium doped tin oxide; Silver nanowires; Flexibility

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INTRODUCTION

As one of the most abundant natural resources, cellulose, is characterized by unique features including renewability, biodegradability, low thermal expansion, and eco-friendly properties (Yan and Xu 2015; Liu *et al.* 2017; Zhang *et al.* 2018, 2019a). This could lead to the development of light weight and low-cost green products. One of these products—cellulose film—is versatile and has a facile chemical modification, smooth surface, controllable film structure, and transparency (Wu *et al.* 2009, 2020; Huang *et al.* 2019; Zheng *et al.* 2019). Furthermore, in terms of functionality, the regenerated cellulose film (RCF) can be optimized from the level of the cellulose molecule, which can prepare tunable and delicate cellulose-based products.

RCF has been employed in several applications (Li *et al.* 2014, 2018; Wu *et al.* 2009). For example, RCF has been employed in the antibacterial packing material, adsorption and catalytic carrier, filter membrane, and even the conductive substrate for photoelectric devices. The conductive and transparent RCF can be supportive in the manufacturing of flat-panel displays, wearable electric devices and sensors, defrosting windows, *etc.* (Cao *et al.* 2014; Lee *et al.* 2014; Zhang *et al.* 2019b). More importantly, transparent RCF presents superior flexibility; thus it could be favorable in the foldable and light-weight electronic devices. Compared with its counterparts, such as glass (Li *et al.*

2018) and plastic (polycarbonate and polyethylene terephthalate) (Lee *et al.* 2014), RCF is flexible, light weight, and eco-friendly.

To prepare conductive RCF, conductive material is introduced onto the surface of RCF, which is required because pure RCF is an insulator. The conductive materials for decorating RCF mainly consist of metals and metal nanowire, carbon-based materials, conductive polymers, and doped metal oxide (Liu *et al.* 2014; Tang *et al.* 2014; Lee and Jeong 2015; Raghunathan *et al.* 2017; He *et al.* 2018; Zhang *et al.* 2019c). Both metals and metal nanowire present excellent ductility and conductivity for the flexible substrate. However, the low transparency (metal luster) for the thin layer in metal, high surface-roughness, and the poor adhesion between the metal nanowire and substrate limit their performance as well as their commercial applications. Carbon-based materials have high transparency and flexibility but possess low electrical conductivity. The carbon-based materials also contain complicated synthesis processes accompanied by environmental issues. The conductive polymer can have a high conductivity with a reduced transparency, especially in the visible wavelength region. Indium-doped tin oxide (ITO) is characterized by excellent optical and electronic properties with high environmental stability. However, due to its brittle nature, ITO has poor compatibility with flexible substrates and is difficult to use in high-throughput roll-to-roll processing. This results in a challenging and urgent problem for constructing the flexible and transparent ITO-based device.

In this paper, ITO and Ag nanowires (AgNWs) were employed as the integrated conductive layer on the RCF surface to prepare flexible, transparent, and conductive RCF. In this conductive layer, the AgNWs formed the conductive framework, and ITO was padded into the AgNWs framework. The advantages of conductivity, flexibility, and transparency were achieved simultaneously through the novel configuration of this conductive layer. The developed technology provides a great potential to accelerate the cellulose diversified applications and high value-added products.

EXPERIMENTAL

Materials

The regenerated cellulose film (RCF) was prepared by the dissolution of softwood dissolving pulp in an 85% N-methylmorpholine-N-oxide (NMMO) aqueous solution. It was regenerated in deionized water. The RCF was 10 μm thick and had a transparency of 93%. Indium tin oxide (ITO) having a target diameter equal to 50 nm (SnO_2 to In_2O_3 equal to 10 to 90) was obtained from Zhongnuoxincai Corporation (Beijing, China). A 0.5% AgNWs-isopropanol suspension was purchased from Nanjing XFNANO Materials TECH Co., Ltd. (Nanjing, China), where the diameter of AgNWs was 100 to 200 nm and the length was 30 to 50 μm .

Preparation of ITO-based Conductive RCF

The ITO-based conductive RCF was prepared by sputtering ITO on the regenerated cellulose film (RCF). The sputtering treatment was carried out from a magnetron sputtering system (TRP-450, Shenyangkexueyiqi Corporation, Shenyang, China), based on a reference procedure (Li *et al.* 2018). First, the RCF and ITO were fixed on the sample table and sputtering table, respectively. There was a spatial distance of 30 cm between them. Subsequently, the vacuum was increased to 6×10^{-4} Pa by using a combined mechanical pump and molecular pump. The working pressure was adjusted to 3.0 Pa with an Ar flow

of 30 sccm at 90 °C. A direct current was employed for sputtering the ITO on the RCF at 120 W power density for a specific time. After the sputtering treatment, the ITO-RCF was held in a zip lock bag in a nitrogen atmosphere.

Preparation of AgNWs Promoted ITO-RCF

The AgNWs was inserted between the ITO and RCF through the spin-coating method. Firstly, diluted AgNWs solutions with different concentrations were prepared by adding the targeted volumes of the original AgNWs solution (30 µL, 50 µL, 70 µL, and 100 µL) into isopropanol (total volume was 200 µL). A 3 by 3 cm² RCF was cast on the spin coater (KW-4A, Institute of Microelectronics of the Chinese Academy of Sciences, Hefei, China) with a dropping of 200 µL diluted AgNWs solution. This was performed with the condition of 1000 rpm in 30 s. The AgNWs-based RCF was dried at room temperature for the subsequent analyses. The ITO was sputtering onto the AgNWs-RCF by following the same procedures described above.

Characterization

The electrical conductivity was determined by using a four-point resistance tester (KDB-3, KunDe Inc., China). The transparency of samples was detected by employing a UV–vis spectrophotometer (Agilent UV-8453, Beijing, China). The morphologies were investigated using a scanning electron microscope (SEM, Hitachi SU8010, Schaumburg, China). The bending resistance of the conductive RCF was indicated by the conductive resistance change. This was indicated after the bending treatment that had a 5 mm radius. The conductive resistance change was calculated according to Eq. 1,

$$\text{Conductive resistance change (\%)} = (R_I - R_0) \div R_0 \times 100\% \quad (1)$$

where the R_0 and R_I were the conductive resistances (Ω per sq) before and after the bending treatment. The flexible performance of the device was tested by using a multichannel physiological signal acquisition and process system (RM6240CD, Chengdu Instrument Factory, China). Two pieces of conductive RCFs were separated by a spacer (a sponge was used to prevent the direct connection of the two RCFs). They were fixed onto the device and encapsulated with copper straps.

RESULTS AND DISCUSSION

Properties of ITO-based Conductive RCF

Figure 1 shows the effect of the sputtering ITO treatment on the conductivity and transparency of the conductive RCF. Pure RCF is an insulator, while the conductivity of RCF can be remarkably improved by the sputtering treatment of ITO. Increasing the treatment time can considerably decrease the conductive resistance of the RCF sample (from Fig. 1). For example, an ITO sputtering in 0.5 h prepared the conductive RCF with 450 Ω per sq from the pure RCF with infinite resistance. By increasing the sputtering time from 0.5 to 1.5 h, the conductive resistance was decreased to 139 Ω per sq and even reduced to 41 Ω per sq after 2.5 h of ITO sputtering. On the other hand, the conductive resistance of RCF was characterized by a non-linear decrease by increasing the sputtering time. From Fig. 1, the resistance decreased rapidly in the beginning, but this reduction occurred slowly in the second phase (even reaching a plateau). This result agreed with the previous reports (Carvalho *et al.* 2005). The effect of ITO sputtering on the transparency at 550 nm

wavelength of RCF is shown in Fig. 1. The ITO sputtering treatment slightly decreased the RCF transparency. This is where a simple linear relation between the RCF transparency and ITO sputtering time was presented during the sputtering process. This can be ascribed to the prolonged sputtering treatment, which resulted in a larger amount and thickness of ITO on the RCF surface. This ultimately reduces the transparency of the conductive RCF. Sahu *et al.* (2018) found that the thickness of the ITO layer affected the conductivity and transparency. In Sahu *et al.* (2018), the resistivity decreased from $1.3 \times 10^{-3} \Omega\text{cm}$, to $6.5 \times 10^{-4} \Omega\text{cm}$, and finally to $5.8 \times 10^{-4} \Omega\text{cm}$. The transparency changed from 82%, to 76%, and finally to 71%. This occurred when the ITO thickness increased from 20 nm, to 30 nm, and finally to 60 nm.

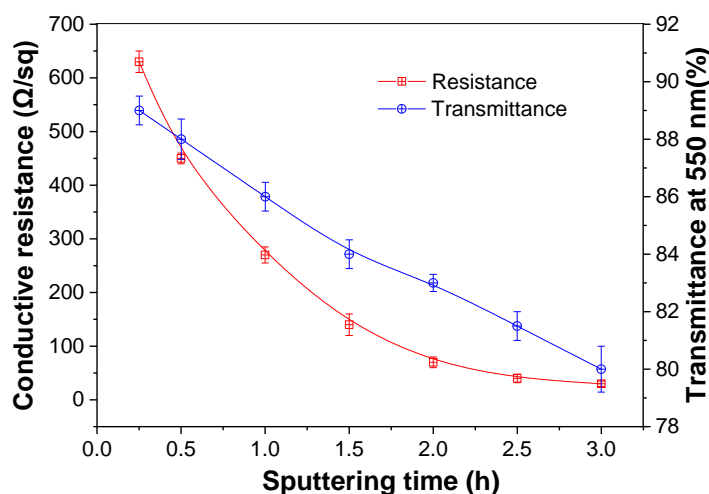


Fig. 1. Conductive RCF as a function of the ITO sputtering treatment

AgNWs Promoted the Properties of ITO-based Conductive RCF

To harvest at a high conductivity/low resistance and high transparency, a 1 h treatment time was the optimized condition for the ITO-RCF (Fig. 1). The subsequent analyses were performed on samples obtained under this condition for 270 Ω per sq of conductive resistance and 86% transparency. In order to further modify the ITO-RCF properties, the AgNWs framework was introduced as the buffer layer between the RCF substrate and the ITO conductive layer. Owing to the excellent electrical conductivity of Ag, a small amount of AgNWs was employed, thus presenting an AgNWs conductive framework on the RCF surface. As shown in Fig. 2, the AgNWs buffer framework considerably improved the conductivity and decreased the sheet resistance of the conductive RCF. An amount of 30 μL of AgNWs decreased the conductive resistance to 210 Ω per sq from 270 Ω per sq of the original ITO-RCF. The resistance was reduced to 170 Ω per sq and 120 Ω per sq from 50 μL and 100 μL of AgNWs, respectively.

Figure 3 shows the effect of AgNWs framework on the transparency of the ITO-based conductive RCF. AgNWs had a negative impact on the transparency of the conductive RCF. When the amount of AgNWs was 50 μL and 100 μL , the transparency of the conductive RCF was 78% and 74%, which was much lower than that of the control conductive ITO-RCF (86%, Fig. 3). Compared with the transparency of conductive ITO-based RCF, the AgNWs showed a more negative effect than the single ITO. For the 140 Ω per sq resistance of the conductive RCF, the transparency was still 84% for the ITO-RCF (Fig. 1) and only 76% of the ITO-AgNWs-RCF (Fig. 2 and 3). Almost 81% of the transparency for the conductive RCF was followed by 40 Ω per sq resistance of the ITO-

RCF (Fig. 1) and 210Ω per sq of the ITO-AgNWs-RCF (Fig. 2 and 3). A larger reduction occurred in the transparency of the conductive RCF due to the AgNWs framework when compared to the single ITO. This can be attributed to the metallic luster of AgNWs (Cao *et al.* 2014), whereas the ITO possesses a relatively high transparency because of its wide band gap. In this studied condition, the AgNWs-based conductive RCF with a $50 \mu\text{L}$ dosage of AgNWs exhibited an 88% transmittance with a 1200Ω per sq conductivity. Mun *et al.* (2016) prepared the strain sensor of AgNWs on cellulose; AgNWs resulted in a low transparency. The transparency was just 70% when the sheet resistance was $2.4 \text{ k}\Omega$ per sq at 0.01 wt\% of AgNWs.

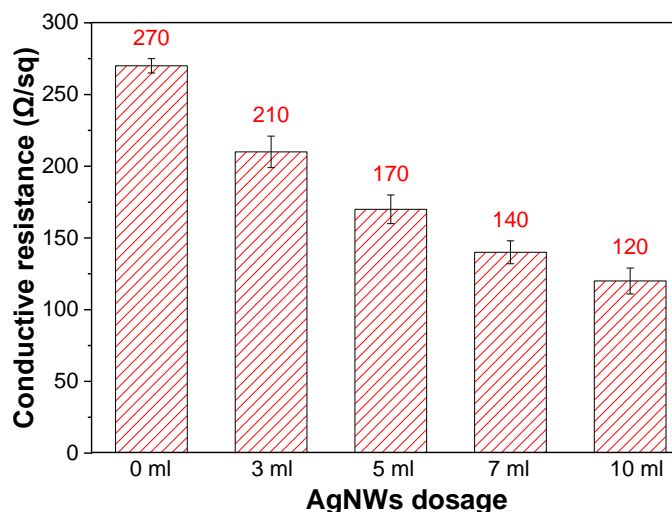


Fig. 2. The effect of AgNWs on the conductivity of the ITO-based conductive RCF

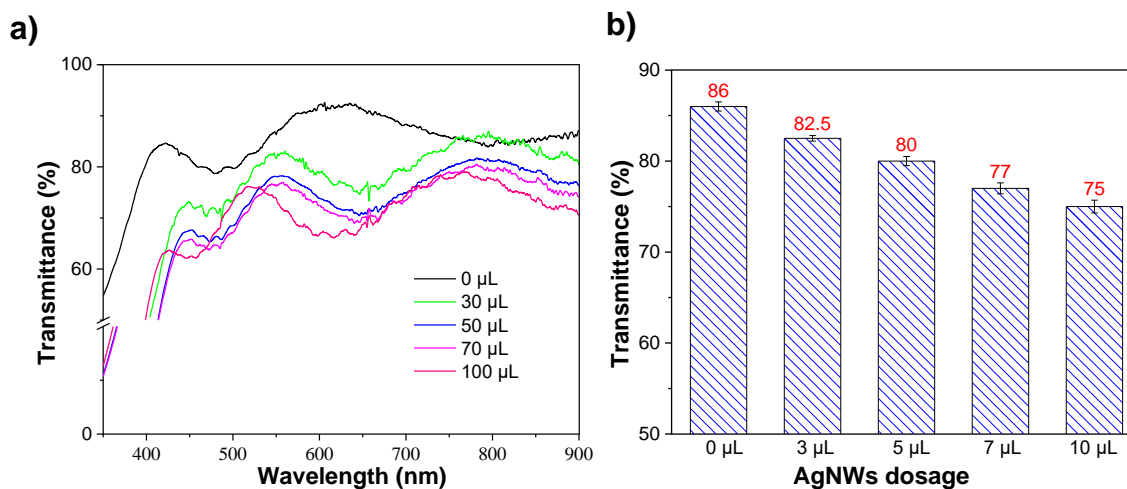


Fig. 3. The effect of AgNWs on the transparency of the ITO-based conductive RCF

Figure 4 presents the SEM observation of the conductive layers on the RCF surface. From Fig. 4a), the ITO layer showed a smooth surface with a roughness of 15 nm, which was located on the RCF. The AgNWs formed a conductive framework due to the small amount and loose deposit on the RCF surface (shown in Fig. 4c) and d). Based on the approach of spin-coating AgNWs followed by depositing ITO on the RCF, an integrated conductive layer clearly formed with the AgNWs framework and ITO filler. It was a desirable configuration for the flexible, transparent, and conductive RCF with low roughness of 30 nm, which will be discussed in the next section.

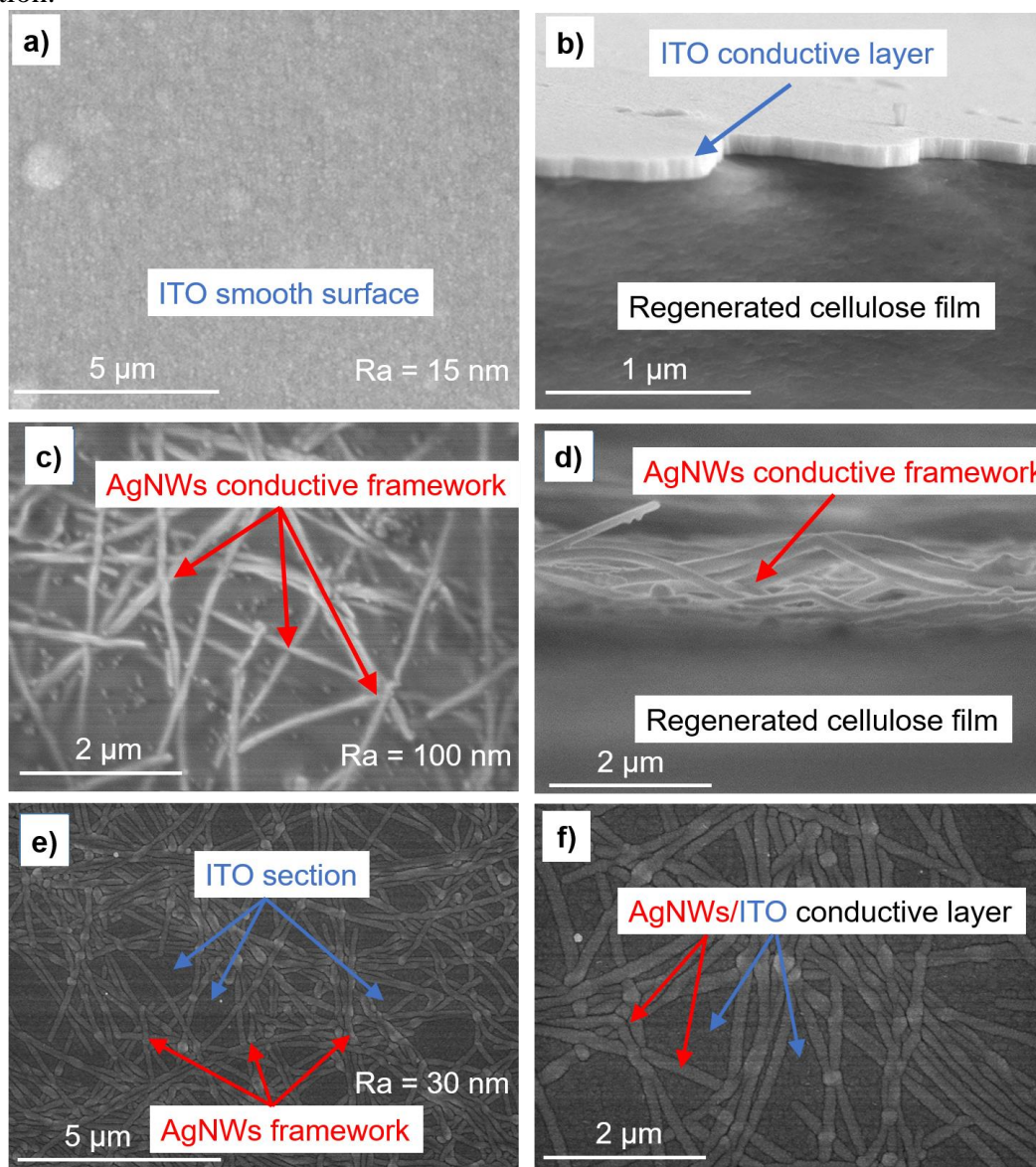


Fig. 4. Morphologies of the conductive RCF: a) ITO surface, b) ITO cross-section, c) 50 μ L AgNWs surface, d) 50 μ L AgNWs cross-section, e) and f) surface of 50 μ L AgNWs framework and ITO filler. R_a : roughness of conductive RCF.

Figure 5 shows the effect of the bending treatment on the resistance of transparent and conductive RCF. The ITO-based conductive RCF without AgNWs had an inferior bending resistance and five times of the bending treatment increased the conductive resistance from 270 Ω per sq to 452 Ω per sq. The resistance was further increased to 1580 Ω per sq by bending 50 times. This indicated the remarkable decrease of conductivity of the ITO-based conductive RCF due to the bending treatment. As the inorganic conductive layer, the ITO was characterized by high rigidity, friableness, and low flexibility (Lee *et al.* 2014; Tran *et al.* 2015). The ITO layer was fractured by the bending treatment, thus disturbing the current carrier mobility. It finally considerably decreased the conductivity of the ITO-based conductive RCF. The results agreed with those reported in the literature that the ITO-based conductive substrates were not suitable for developing flexible conductive devices (Cao *et al.* 2014; Li *et al.* 2018).

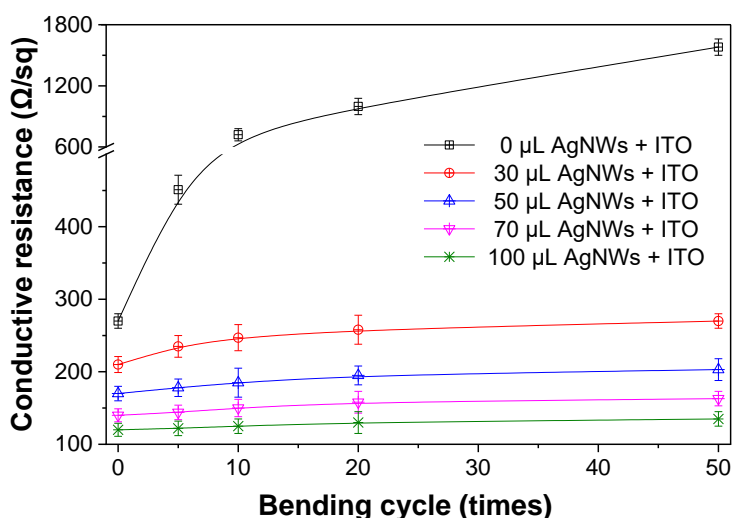


Fig. 5. The effect of AgNWs framework on the bending capacity of conductive ITO-RCF

AgNWs present outstanding toughness and flexibility (Cao *et al.* 2014; Lee *et al.* 2014). These features can be favorable for the bending resistance of ITO-based conductive RCF. Figure 5 presents that with the addition of 30 μL AgNWs, 5 times of bending treatment only increased the conductive resistance from 210 Ω per sq to 235 Ω per sq (increased 25 Ω per sq, much lower than 182 Ω per sq of single ITO-RCF, from 270 Ω per sq to 452 Ω per sq). 50 times of bending resulted in an increase of 60 Ω per sq from 210 Ω per sq, to 270 Ω per sq. The bending resistance of the conductive RCF can be further modified due to increasing the amount of AgNWs. In relation to the bending treatment of 50 times, the resistance change of the conductive RCF was 33 Ω per sq for 50 μL AgNWs (from 170 Ω per sq to 203 Ω per sq), while only 15 Ω per sq for 100 μL of AgNWs (from 120 Ω per sq to 135 Ω per sq), respectively. One study (Hu *et al.* 2010) has demonstrated less of an increase in sheet resistance for the AgNWs than in ITO by the bending treatment. The results went from 17 Ω per sq to 21 Ω per sq and from 51 Ω per sq to 6 k Ω per sq after 100 times of bending in 5 mm. Sannicolo *et al.* (2016) also elucidated that the AgNWs presented a more negligible increase in electrical resistance for bending and better responses of stretching than the ITO. The AgNWs was favorable for manufacturing flexible electrodes.

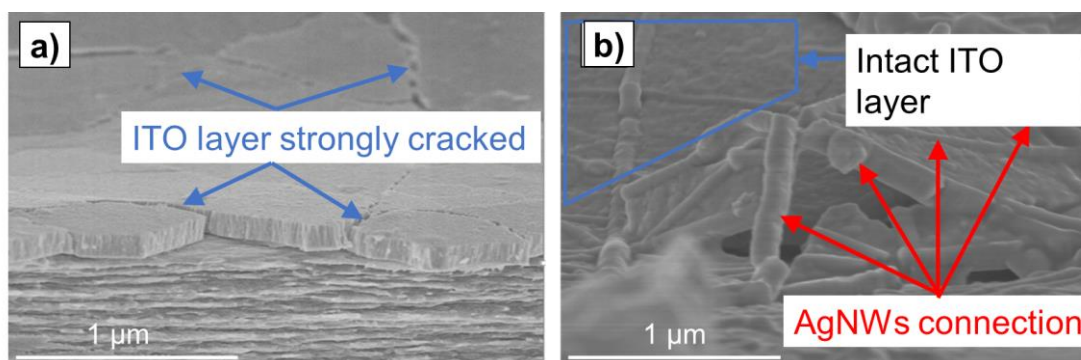


Fig. 6. Morphologies of the conductive RCF after the bending treatment: a) ITO layer, b) 50 μ L AgNWs framework and ITO filler

The effect of the bending treatment on the conductive RCF morphology is presented in Fig. 6. Due to the fragile nature of ITO, the bending treatment remarkably destroyed and cracked the ITO conductive layer. The bending treatment even led to the peeling of the ITO from the RCF (Fig. 6a). The resultant fracture of the ITO layer hindered the electron transportation, thus considerably decreased the conductivity of ITO-based conductive RCF (largely increasing its conductive resistance in Fig. 5) due to the bending treatment. As shown in Fig. 6b, the bending treatment presented a weaker effect on destroying and deconstructing the ITO layer for the AgNWs-ITO sample than that of the ITO's single sample. This was because the AgNWs can absorb and transfer the bending stress. As a result, it was desirable to protect the fragile ITO conductive layer. Moreover, although the integrated AgNWs-ITO conductive layer was partial destroyed, the flexible AgNWs were favorable for linking and connecting. The AgNWs framework was also supportive for the electron transportation in the integrated conductive substrate. Based on this, the AgNWs-ITO conductive RCF presented a lower loss of conductivity than the control ITO based RCF after the bending treatment.

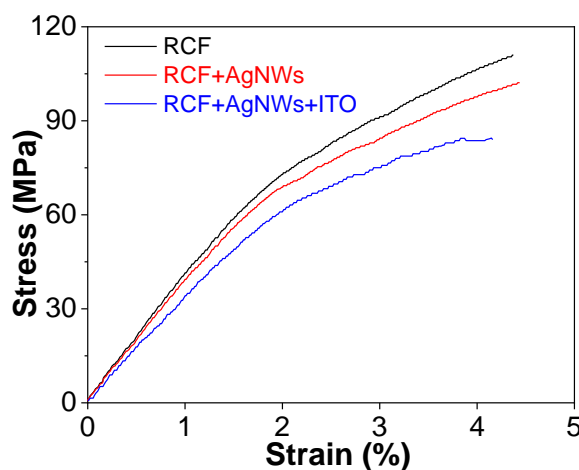


Fig. 7. The strength of the conductive RCF. 50 μ L AgNWs surface; 1 h sputtering time of ITO

The mechanical strength of conductive regenerated cellulose film with AgNWs and ITO was investigated (Fig. 7). The original RCF showed a tensile strength of about 110 MPa with a strain at break of 4.4%. The conductive RCF with AgNWs presented a similar stress-strain curve with the original RCF. When introducing ITO onto the RCF, the strength of conductive regenerated cellulose film was slightly weaker than the original cellulose film. This may be due to the fact that ITO can insert itself into the inner structure of cellulose film, thus disturbing the bonding of cellulose molecules.

Highly Flexible Devices Based on Conductive Regenerated Cellulose Film

The flexible and conductive RCF with AgNWs interface modification was employed for constructing a simple electronic device. The high flexibility of the RCF-based device was further demonstrated by investigating the effect of the bending treatment on the conductivity performance of the devices. When there was no pressure on the device, the electric circuit was in the off state with the indication of enough high resistance, as shown in Fig. 8a). Bending the device can achieve the connection of the conductive substrates; thus a superior electric circuit was built (Fig. 8b). Figure 8 further describes the flexibility performance of the electric device.

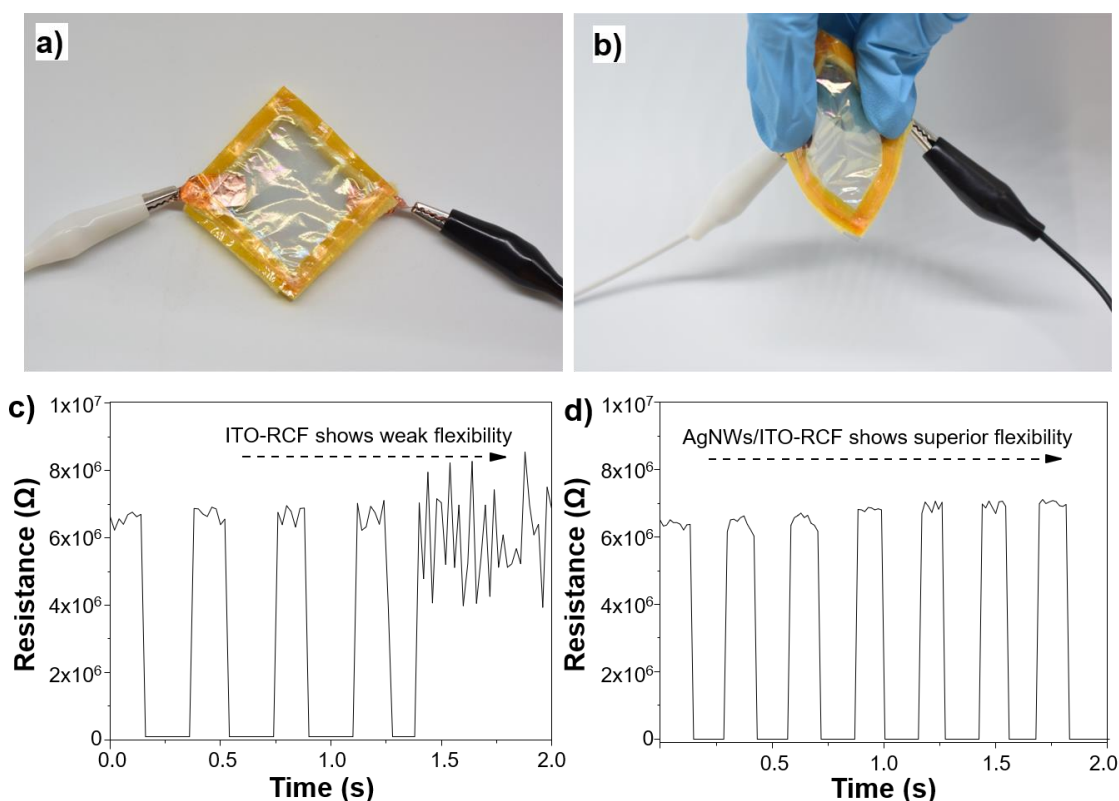


Fig. 8. Flexibly electric device: a) Commonly, the electric device was in the off state with the indication of sufficiently high resistance due to the non-connection of the two RCFs, b) Bending the device can achieve the connection of the conductive substrates, thus build an electric circuit, c) The control ITO-RCF showed weak performance relative to bending, due to the conductivity decrease after bending, d) AgNWs modified ITO-RCF illustrates an outstanding flexibility, where a regular change of conductive resistance appears when bending and releasing the device

The control ITO-RCF based device illustrated a strong decrease in the conductive performance and even exhibited non-conductivity after several bending treatments. This was a response to the weak flexibility of this kind of device due to the rigid and brittle feature of the ITO (Fig. 8c). However, the AgNWs interface modification endowed the RCF-based electric device with high flexibility and toughness. This feature was indicated by the regular change of resistance, thus was in an insulated state when no connection existed in the conductive RCF. The conductive RCF was in a conductive state of 250 Ω during bending (Fig. 8d).

CONCLUSIONS

1. Conductive RCF was developed with the superior features of the high conductivity (170 Ω per sq) and transparency (78%). This was accomplished by the layered growth of spin-coating the AgNWs and sputtering the ITO.
2. Based on the cooperation of AgNWs and ITO, the conductive RCF showed desirable flexibility, where the conductivity was much lower than the control conductive RCF after the bending treatment.
3. The conductive, transparent, and flexible RCF was successfully used to construct a flexible device with a high sensitivity and tolerance.

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